



Ultrafast Light-Driven Electronic and Structural Changes in LaFeO₃ Perovskites Probed by Femtosecond X-Ray Absorption Spectroscopy

Masoud Lazemi,* Fabian J. Mohammad, Sang Han Park, Abhishek Katoch, Hans J.F.A. Blankesteijn, Andrés R. Botello-Méndez, Emma van der Minne, Yorick A. Birkhölzer, Iris C. G. van den Bosch, Ellen M. Kiens, Christoph Baeumer, Gertjan Koster, Soonnam Kwon, Uwe Bergmann, and Frank M. F. de Groot*

Conducting real-time, element-specific studies of photo-excited systems is a long-standing challenge. The development of X-ray free-electron lasers (XFELs) has paved the way for the emergence of a promising technique: femtosecond X-ray absorption spectroscopy (fs-XAS). This powerful technique reveals electronic and geometric characteristics, providing unprecedented insight into their dynamic interactions under nonequilibrium conditions. Herein, the fs-XAS technique is employed at PAL-XFEL to unravel light-driven ultrafast electronic and structural changes in epitaxial lanthanum iron oxide (LaFeO₃) thin films. Density functional theory (DFT) and multiplet calculations are utilized to expound on the experimental results. The analyses reveal that photoexcitation initially induces high- and intermediate-spin Fe²⁺ states through ligand-to-metal charge transfer (LMCT), followed by polaron formation. It is demonstrated that the reduced overlap between the oxygen 2p and iron 3d orbitals accounts for all experimental observations, including 1) the XAS shifts to lower energies, 2) the decrease in the crystal field splitting, and 3) the relatively larger shifts observed in the oxygen 1s XAS.

1. Introduction

Developing advanced photoanode materials with tailored properties is crucial for enhancing photoelectrochemical water splitting. Titanium dioxide (TiO₂) and hematite (α -Fe₂O₃), thus far,

M. Lazemi, F. J. Mohammad, H. J. Blankesteijn, A. R. Botello-Méndez, F. M. F. de Groot

Materials Chemistry and Catalysis

Debye Institute for Nanomaterials Science

Utrecht University

Universiteitsweg 99, Utrecht 3584 CG, The Netherlands E-mail: m.lazemi@uu.nl; F.M.F.deGroot@uu.nl

D The ORCID identification number(s) for the author(s) of this article can be found under https://doi.org/10.1002/adma.202502932

© 2025 The Author(s). Advanced Materials published by Wiley-VCH GmbH. This is an open access article under the terms of the Creative Commons Attribution License, which permits use, distribution and reproduction in any medium, provided the original work is properly cited.

DOI: 10.1002/adma.202502932

tion spectroscopy (XAS) accessible for time-dependent studies on the femtosecond (fs) time scale (fs-XAS).^[3-9] A wide range of intriguing applications beyond static synchrotron experiments are heralding the dawn of a new era in many scientific disciplines.^[10] Our group has recently reported the hole transport dynamics in TiO₂ in a direct and real-time

have been the most studied metal oxides

as photoelectrodes.^[1] However, their large

bandgaps and extremely short charge carrier diffusion lengths pose signif-

icant challenges to achieving higher

solar-to-hydrogen conversion efficien-

cies. Lanthanum transition-metal oxides

(LaTMO₃) have garnered considerable

attention as rising candidates.^[2] Vary-

ing the TM enables the creation of a myriad of lanthanum-based perovskites

with distinct chemical and physical prop-

erties as well as photoelectrochemical

performance. Understanding and opti-

mizing the performance of such mate-

rials necessitate advanced spectroscopy techniques capable of probing their ultra-

fast electronic and structural dynamics.

lasers (XFELs) has made X-ray absorp-

The emergence of X-ray free-electron

M. Lazemi, E. van der Minne, Y. A. Birkhölzer, I. C. G. van den Bosch, E. M. Kiens, C. Baeumer, G. Koster MESA+ Institute for Nanotechnology University of Twente P.O. Box 217, Enschede 7500 AE, The Netherlands S. H. Park, S. Kwon Pohang Accelerator Laboratory Pohang, Gyeongbuk 37673, South Korea A. Katoch Department of Chemistry Yonsei University Seoul 03722, South Korea U. Bergmann Department of Physics University of Wisconsin-Madison Madison, WI 53706, USA

ADVANCED SCIENCE NEWS ______ www.advancedsciencenews.com ADVANCED MATERIALS



Figure 1. a) The pump-probe XAS experimental setup at the Soft X-ray Scattering and Spectroscopy (SSS) beamline of PAL-XFEL b) Schematic of the LaFeO₃ perovskite structure.

manner.^[4] Besides TiO₂, we have reported various fs-XAS studies on photoelectrocatalytic materials, including MoTe₂, α -Fe₂O₃, CeO₂, and CuWO₄.^[3,11-14]

Despite extensive studies on the photocatalytic activity and ground-state properties of lanthanum iron oxide (LaFeO₃), prior investigations have primarily relied on static spectroscopic or electrochemical techniques, lacking direct insight into carrier dynamics and excited-state processes.^[15–19] Such approaches are unable to capture the ultrafast processes and transient electronic structures that underlie key efficiency-limiting mechanisms, including carrier recombination and polaron formation. In particular, the lack of element- and orbital-specific time-resolved measurements has hindered a detailed understanding of charge localization and spin state evolution following photoexcitation.

Herein, we focus on the fs-XAS study of photoexcited LaFeO₃ epitaxial thin films at the PAL-XFEL.^[20] We and other groups have reported the promising activity of LaFeO₃ for the oxygen evolution reaction as a photoanode.^[15,18] However, to the authors' knowledge, the dynamics of photo-excited carriers in LaFeO₃ have not yet been explored. After discussing the ground state (GS), excited state (ES), and transient (TR) XAS spectra, as well as kinetic traces at the oxygen 1s (K) edge of LaFeO₃, we employ density functional theory (DFT) calculations to interpret the spectra. We then analyze similar spectra at the iron $2p_{3/2}$ (L₃) edge and corroborate them with multiplet calculations. We demonstrate how photoexcitation initially induces high- and intermediate-spin Fe2+ states through ligand-to-metal charge transfer (LMCT) following polaron formation. Our quantitative analysis of the observed fs-XAS spectra provides detailed insights into the mechanisms underlying

the dynamics that follow the photoexcitation of this promising photoelectrocatalyst.

2. Results and Discussion

Figure 1a,b depict the experimental setup used in our pumpprobe fs-XAS experiment to investigate the photoexcited state of LaFeO₃, which has an optical band gap of \approx 2.0 eV.^[21] The sample was excited by a 400 nm (3.1 eV) optical pump laser with an 80-fs pulse width, followed by probing at the oxygen 1*s* (K) and iron 2*p*_{3/2} (L₃) edges, using 80-fs X-ray pulses.

2.1. Ground-State and Transient Oxygen 1s XAS Spectra

2.1.1. DFT Calculations

The ground-state (GS) oxygen 1s XAS spectrum of LaFeO₃ is depicted in **Figure 2a**. The shape of the oxygen 1s XAS can be approximated by the unoccupied states with the oxygen *p* symmetry in the presence of a core hole.^[22,23] We have performed DFT calculations to interpret the XAS spectra, as shown in Figure 2d,e. LaFeO₃ has a $3d^5$ high-spin $t_{2g_1}^3 e_{g_1}^2$ (${}^6A_{1g}$) configuration, suggesting that the unoccupied iron 3d spectral density will mainly have t_{2g_1} and e_{g_1} characters.^[24–26] Hence, in Figure 2a, the features at 529.86 eV (hereafter Peak B) and 531.45 eV are attributed to transitions into iron t_{2g_1} and e_{g_1} bands, respectively.

The t_{2g} and e_g peaks are assigned to hybridization between oxygen 2p and iron 3d states. It should be noted that there is a







Figure 2. Comparative analysis of oxygen 1s fs-XAS of LaFeO₃ a) ground state (GS) and excited state (ES) spectra at 0.2, 1, and 5 ps. The energy shift for the t_{2g} and e_g peaks are -0.7 and -1 eV, respectively. The ES spectra were derived by Equation S5 (Supporting Information), *ES* (λ , t) = $\frac{1}{\alpha}$ *TR*(λ , t) + *GS*(λ , 0) > 0, considering α = 0.1. The inset shows the enlarged region between 526.5 and 529.5 eV. The transient peak, marked by an arrow, is attributed to the ultrafast transition of oxygen 1s electrons to oxygen 2*p* hole states. Similar spectra with different α values are plotted in Figure S3 (Supporting Information). b) The 1st derivative of GS and transient (TR) spectra at 0.2, 1, and 5 ps. For clarity, the TR spectra were multiplied by 5, while the 1st derivative of GS was divided by 3. c) Kinetic traces of transient XAS at Peak A (526.86 eV), Peak B (529.86 eV), and Peak C (533.46 eV), fitted using Equation S6 (Supporting Information). The error bars are depicted as vertical lines. d) Calculated electronic band structure of LaFeO₃ along high-symmetry points e) Calculated projected density of states (PDOS) of LaFeO₃ as a function of E – E_F, depicting the total density of states (TDOS), lanthanum 5*d* and 5*f*, iron 3*d*, and oxygen 2*p* states.

subtle difference between the t_{2g} and e_g splitting compared with crystal field splitting (CFS). The energy difference between the t_{2g} and e_g at the oxygen 1s XAS was found to be 1.6 eV (Figure S1 and Table S1, Supporting Information), which agrees well with previous reports.^[27–29] Nonetheless, this value is smaller

than CFS resulting from iron 2*p* XAS calculations, i.e., 1.8 eV, as reported in Section 2 and elsewhere.^[24] The underlying reason for these differences is that the oxygen 1*s* core hole reduces the splitting between the two peaks. In iron oxide compounds, the core hole potential pulls down the t_{2g} and e_g states to the

ADVANCED MATERIALS

bottom of their bands. The e_g band is broader than the t_{2g} band, and the energy difference between the centers of the t_{2g} states and e_g states is reduced to a value of 1.6 eV in the GS oxygen 1s XAS. The same effect is visible in Fe₂O₃, where the average $t_{2g}-e_g$ splitting is reduced to a 1.2 eV splitting in the oxygen 1s edge.^[23,30,31]

In Figure 2d,e, we identify three distinct regions by setting the Fermi level at the top of the valence band (VB):

- Region I: dominant contributions from the oxygen 2*p* orbitals (between -6 and 0 eV)
- Region II: antibonding iron 3*d* and oxygen 2*p* states (between 2 and 4.5 eV)
- Region III: antibonding lanthanum 5*d* and oxygen 2*p* states (between 5 and 10 eV)

The band structure between 2.0 and 2.5 eV is almost nondispersive across momentum space due to the localized nature of the unoccupied Fe³⁺ minority spin t_{2g} states. The states between 2.5 and 4.5 eV are primarily derived from the minority spin iron 3*d* states, and their overlap with the oxygen 2*p* states results in a larger bandwidth and dispersion than for the t_{2g} states.

2.2. Excited State, Transient Spectra, and Kinetic Traces

2.2.1. Region I: Oxygen 2p States

We acquired the kinetic traces below the oxygen 1s threshold energy at Peak A (526.86 eV) to examine the behavior of holes upon photoexcitation, as depicted in Figure 2a,c. According to our DFT calculations in Figure 2d,e, Peak A resides in Region I, with dominant contributions from the oxygen 2p orbitals. This assignment is consistent with earlier DFT-based theoretical studies on similar transition-metal oxides, which attribute pre-edge features in oxygen K-edge XAS to oxygen 2p states.^[21,29]

In this energy range, the oxygen 2p orbitals are not only involved in metal–oxygen bonding but also play a significant role in forming ligand-hole states upon photoexcitation. These hole states are signatures of LMCT, where an electron from oxygen 2p is transferred to an iron 3d state. This is further supported by the transient spectrum (see inset of Figure 2a), which shows a distinct increase in intensity at Peak A shortly after excitation. This transient feature indicates an increase in the unoccupied oxygen 2p character near the valence band edge, consistent with the formation of photoinduced holes.

This interpretation is in line with our previous work on α -Fe₂O₃, where we assigned a similar pre-edge feature at 527.6 eV to transitions involving oxygen 2*p* states.^[12] In that system, we observed a decay constant of $\tau_{decay} = 200$ fs, which is attributed to rapid hole recombination. In contrast, the kinetic trace in Figure 2c for LaFeO₃ reveals a slower decay time of $\tau_{decay_A} = 400$ fs, indicating longer-lived hole states and suggesting more favorable charge separation in LaFeO₃. The longer hole lifetime in LaFeO₃ implies a more significant opportunity for the photogenerated holes to participate in oxidation reactions, thereby enhancing its potential as a photoanode material for the oxygen evolution reaction (OER). The kinetic constant for the rising (decaying) region of Peak A (Peak B) around time zero provides the instrumental resolution, which is 120 fs (Figure 2c). This aligns well

with the resolution that we previously achieved for TiO_2 , Fe_2O_3 , $CuWO_4$, and CeO_2 at the same beamline.^[4,11–13,32]

2.2.2. Region II: Antibonding Fe 3d and Oxygen 2p States

The oxygen 1s XAS in Region II manifests the unoccupied iron 3d orbitals. In other words, XAS is proportional to the oxygen character of the unoccupied iron 3d orbitals.^[22,23] In LaFeO₃, each iron atom is octahedrally coordinated by six oxygen atoms (Figure 1b), resulting in the crystal field splitting of iron 3d orbitals into $t_{2\sigma}$ and e_{σ} states, as expected in an octahedral ligand field. As discussed in Section 1.1, Peak B and the neighboring features in Region II are attributed to these anti-bonding t_{2g} and e_g orbitals. Specifically, Peak B corresponds predominantly to hybridized iron 3d t_{2g} -oxygen 2p states, while the shoulder at higher energy reflects transitions into e_{o} -derived states. These assignments are supported by our calculated projected density of states (PDOS) results (Figure 2e), where the iron 3d density splits into two sub-bands consistent with the $t_{2\sigma}$ and e_{σ} separation. Previous studies employing oxygen K-edge XAS have reported comparable orbital assignments in perovskite oxides.[22,23]

The comparison of ground state (GS) and excited state (ES) spectra in Figure 2a,b reveals two key spectral changes in Region II following photoexcitation:

- A redshift of both t_{2g} and e_g peaks (-0.7 and -1.0 eV, respectively), indicative of changes in the local crystal field and bonding environment.
- A reduction in XAS intensity at these peaks can be attributed to the population of unoccupied iron 3d bands by photoexcited electrons. The redshift of the t_{2g} and e_g peaks indicates that the corresponding antibonding states in the excited state (ES) are less anti-bonding than in the ground state (GS). This suggests an increased contribution from iron 3d orbitals and a reduced contribution from oxygen 2p orbitals in these bands. As the oxygen 2p character decreases, the transition probability at the oxygen K-edge diminishes, resulting in a lower XAS intensity.

2.2.3. Region III: Antibonding La 5d and Oxygen 2p States

Region III encompasses Peak C (533.46 eV), which is assigned to lanthanum 5*d*–oxygen 2*p* antibonding states (Figure 2d,e). The observed rise in Peak C following photoexcitation in Figure 2c, with a kinetic time constant of ≈ 600 fs, suggests that photoexcitation induces a delayed modification of the lanthanum–oxygen bonding environment. This occurs relative to the faster hole dynamics in the oxygen 2*p* and iron 3*d* orbitals (Region I). This delay, similar to the 400-fs decay time at Peak A, reflects structural relaxation or polaron formation.

Importantly, no significant transient changes are observed in the lanthanum 3*d* fs-XAS spectra (Figure S4, Supporting Information), indicating that the lanthanum 4*f* and core states remain unaffected by photoexcitation. This is consistent with the known weak hybridization between La 4*f* states.^[29] The La atoms act primarily as "spectators," and the observed dynamics in Region III are better understood as an indirect consequence of changes in the iron–oxygen sublattice rather than direct excitation of lanthanum orbitals.





Figure 3. Comparative analysis of iron $2p_{3/2}$ fs-XAS of LaFeO₃ a) ground state (GS) and excited state (ES) spectra at the delay times of 0.2, 1, and 5 ps. The sticks represent crystal field multiplets calculated by CTM4XAS. The GS spectrum was calculated using 10Dq = 1.8 eV. The t_{2g} and e_g peaks shift -0.1 and -0.4 eV, respectively. The ES spectra were derived using Equation S5 (Supporting Information), considering $\alpha = 0.1$. The similar ES spectra with different α values are plotted in Figure S7 (Supporting Information). b) Kinetic traces of transient XAS at Peak D (706.5 eV) and Peak E (708.7 eV). c) Kinetic traces of transient XAS at Peak F (705.5 eV) and Peak G (707.5 eV). The error bars are depicted as vertical lines. d–f) Experimental and simulated transient (TR) spectra at the delay times of 0.2, 1, and 5 ps, considering Fe²⁺ high-spin (HS), Fe³⁺, polaron Fe³⁺, and Fe²⁺ low-spin (LS).

2.3. Iron 2p_{3/2} fs-XAS Spectra

In the 2*p* XAS process, each $3d^N$ state is excited to a final $2p^5 3d^{N+1}$ state.^[22] Our previous works include several comprehensive reviews of 2p XAS of 3d transition metal systems in refs. [30,33–35]. These transitions are strongly influenced by both core-hole effects and multiplet interactions arising from the higher-order electron-electron interactions between the 2p core hole and the 3d valence electrons. In general, in the iron 2p XAS spectrum of LaFeO₃, the separation between the two peaks results from the final-state spin-orbit splitting of the core 2p states to $2p_{3/2}$ and $2p_{1/2}$ peaks from where the electrons are excited.^[22] Note that, in the initial state, these states are degenerate. As depicted in **Figure 3**a, we focus on the $2p_{3/2}$ region, which itself exhibits further splitting due to the crystal field interaction-the splitting of Fe 3*d* orbitals into t_{2g} and e_g levels under the octahedral coordination of oxygen atoms. This crystal field splitting (CFS) manifests as distinct spectral features in the iron $2p_{3/2}$ XAS and serves as a sensitive probe of the local electronic structure and oxidation state of iron.

The iron 2p XAS spectra can be simulated by calculating the crystal field multiplets in octahedral symmetry.^[22,36] We have developed a theoretical approach based on multiplet calculations to comprehend the time evolution of the iron $2p_{3/2}$ XAS spectra. This approach explicitly incorporates spin-orbit coupling, crystal field effects, and LMCT. Figure 3d-f illustrates the calculated transient (TR) spectra for different delay times: 0.2, 1, and 5 ps. The GS is $t_{2g\uparrow}^3 e_{g\uparrow}^2 ({}^6A_{1g})$ with all the same spin direction. Therefore, the 6th excited 3*d* electron is spin-down. The final state is thus $t_{2g\uparrow}^3 e_{g\uparrow}^2 + t_{2g\downarrow}^1 + \underline{L}$, where \underline{L} denotes a ligand-hole. The presence of the ligand hole reflects the hybridized nature of the excited state, resulting in new electronic transitions and spectral redistribution. Moreover, photoexcitation leads to a parity inversion. Since the $3d^5$ ground state has odd parity, the optically excited 3d⁶L configuration is of even parity and thus differs from the odd-parity 3d⁶L states that contribute to the ground state.

We will now discuss the observations at the iron $2p_{3/2}$ edge reported in Figure 3.

www.advmat.de

2.3.1. Ground State (GS)

The GS spectrum in Figure 3a features notably sharp multiplets and can be theoretically simulated by assuming a high-spin $t_{2g\dagger}^3 e_{g\dagger}^2$ (⁶ A_{1g}) ground state with 10Dq = 1.8 eV.^[24] This indicates a stable, well-defined structure where the iron–oxygen bond lengths and the surrounding lattice are in equilibrium. A comparison between the experimental and calculated GS spectra (Figure S5, Supporting Information) validates this assignment, showing that the GS of LaFeO₃ is mainly $3d^5 (t_{2g\dagger}^3 e_{g\dagger}^2)$ high-spin, consistent with our previous findings.^[11]

2.3.2. Immediate Response (0.2 ps)

At 0.2 ps after photoexcitation, the system enters a highly nonequilibrium state, where rapid electronic redistribution alters the local electronic environment. The reduction in crystal field splitting (CFS) from 1.8 to 1.5 eV (Figure 3d) is indicative of a weakened ligand field strength due to transient changes in ironoxygen orbital overlap. This is a direct consequence of the excitation of electrons from oxygen 2*p*-derived valence band states to iron 3*d* conduction band states, resulting in LMCT. The presence of LMCT-driven high-spin and intermediate-spin Fe²⁺ states is supported by the observed transient spectral features at 0.2 ps, with a characteristic decay constant of 350 fs (Figure 3c).

In addition, the emergence of a distinct positive feature at 707.5 eV is attributed to the formation of Fe³⁺ polaron states, which are localized states arising from lattice distortion around the iron site. Polaron formation is typically accompanied by a local lattice expansion and electron self-trapping, which can significantly influence carrier mobility and lifetime.^[37] Such behavior has been widely reported in related transition-metal oxides.^[12,13,38] Therefore, the early-time dynamics involve both electronic transitions and coupled electron–phonon interactions, which contribute to the observed spectral modifications.

2.3.3. Intermediate Relaxation (1 ps)

At 1 ps, the system begins to relax toward its original configuration. The CFS increases to 1.75 eV (Figure 3e), suggesting partial restoration of iron–oxygen orbital overlap and local symmetry. Most high-spin and intermediate-spin Fe²⁺ states have decayed by this time, as evidenced by the diminished intensity \approx 705.5 eV. The remaining spectral intensity is attributed predominantly to non-excited Fe³⁺ centers in a nearly relaxed environment. The partial recovery of the crystal field suggests that some electron-hole recombination and structural reordering have occurred, though the system has not yet fully returned to equilibrium.

2.3.4. Long-Term Relaxation (5 ps)

By 5 ps, the system exhibits a CFS of 1.8 eV (Figure 3f), nearly identical to the ground state value, indicating that the electronic and structural relaxation is almost complete. Most photoexcited carriers have recombined, and the lattice distortions introduced

by excitation have largely subsided. The calculated spectra suggest that the dominant residual signal at 707.5 eV corresponds to polaronic Fe^{3+} states, which persist even at later times.

In Figure 3b, the observed behavior of Peak D (t_{2g} -related) and Peak E (e_g -related) can be explained by the distinct dynamics of the electronic states and their relaxation pathways.

2.3.5. Initial Sub-Picosecond Decay

Both the t_{2g} and e_g features exhibit a rapid intensity drop on the sub-picosecond timescale, with decay constants of ≈ 800 fs for Peak D and 900 fs for Peak E. This initial decay phase is attributed to the ultrafast relaxation of photoexcited charge carriers, specifically electrons in iron 3d orbitals and holes in oxygen 2p orbitals. The fast decay is most likely driven by strong electron–phonon coupling, where the excess energy from electronic excitation is rapidly transferred to the lattice via phonon emission. Since both t_{2g} and e_g orbitals participate in bonding and share similar orbital symmetries with the oxygen ligands in the octahedral field, their initial coupling to phonons may be comparable, resulting in similar early-time decay constants.

2.3.6. Long-Term Decay Dynamics

In contrast, the long-time behavior diverges significantly; Peak D, related to the t_{2g} state, decays with a time constant of 100 ps, while Peak E, related to the e_g state, persists for a much longer duration, with a decay constant of 850 ps. This pronounced difference suggests distinct relaxation pathways and decoupled long-term dynamics for the two orbital manifolds. The distinct behavior of Peaks D and E reflects the orbital-dependent nature of the relaxation processes, demonstrating how different iron 3d states interact with lattice, spin, and orbital degrees of freedom on ultrafast and longer timescales.

After the initial fast relaxation, the remaining excited $t_{2\sigma}$ electrons may quickly relax back to the ground state by coupling to lattice vibrations (phonons) or through non-radiative recombination processes. The significantly longer decay time for the e_a peak suggests that excited e_g electrons are less efficiently coupled to the lattice and may require a longer relaxation time. This could be due to weaker electron-phonon coupling for the e_g states or the involvement of more complex relaxation processes. These could include trapping carriers in metastable states or slower recombination dynamics. Alternatively, the longer e_g lifetime could also be due to slower relaxation pathways associated with spin or orbital relaxation processes, as the e_g states might exhibit a different interaction with the spin or orbital degrees of freedom compared to t_{2g} states. These observations could stem from other effects, such as metastable state formation and spin-orbital dynamics discussed in Section 5.3 (Supporting Information).^[39-42]

2.4. Shifts and Changes in the XAS Intensities

Figure 4 depicts the energy diagrams for the ground state and the excited state of $LaFeO_3$ observed in Figures 1a and 3a. The final state of the iron 2*p* XAS includes an iron 2*p* core hole, which introduces an additional potential on the iron 3*d* states. This added



Figure 4. Energy diagrams for LaFeO₃ a) ground state b) the energy shifts in the excited state, with -0.4 and -0.7 eV, for the t_{2g} band and the e_g band, respectively. The iron 2p XAS has reduced shifts (e = -0.3 eV), while the oxygen 1s XAS has increased shifts (e = 0.3 eV). As a result of the decreased overlap between the iron and oxygen orbitals, the valence band becomes increasingly dominated by the oxygen 2p character, while the iron 3d band is more strongly characterized by the iron 3d orbitals.

potential localizes the iron 3*d* orbitals, reducing their overlap with the oxygen 2*p* states. As a result, the anti-bonding states become less anti-bonding and shift to lower energies (Figure 4b). This effect is more pronounced in the e_g states, which are more anti-bonding, thereby reducing the CFS between the t_{2g} and e_g levels. Although the iron 2*p* XAS is strongly influenced by the multiplet effects, the final-state CFS is known to be $\approx 20\%$ smaller than the value observed in optical measurements, bringing it down to $\approx 1.8 \text{ eV}.^{[43]}$

The final state of the oxygen 1s XAS includes an oxygen 1s core hole that acts as an extra potential to the oxygen 2p states. This effect is similar to the influence of the iron 2p core hole on the iron 3d states, but it introduces an additional impact. In essence, the core hole potential also affects the empty 3d states, deforming the band structure and piling up the intensity at the bottom of each band (Figure 1e). Due to the sharp profile of the t_{2g} band, its peak energy decreases only slightly after lifetime broadening. In contrast, the broader e_g band experiences a more significant shift as the PDOS piles up at the bottom of the band. Together, these effects reduce the $t_{2g}-e_g$ splitting by an additional 0.2 eV (with respect to the iron 2p core hole), bringing it down to 1.6 eV (Figure 2a).

In short, the experiment shows an e_{σ} band shift of -1.0 eV and a $t_{2\sigma}$ band shift of -0.7 eV for the oxygen 1s XAS (Figure 2a) upon excitation by a 400 nm laser. In the case of iron 2p XAS, the e_g band shifts -0.4 eV, and the t_{2g} band shifts -0.1 eV (Figure 3a). These shifts can be understood as follows: An optical excitation transfers an electron from the oxygen 2p VB to the iron 3d band and reduces the overlap between the iron and oxygen orbitals, causing less bonding and less anti-bonding mixed states. This implies that both the t_{2g} and e_g bands shift to lower energies than the ground state, as indicated in Figure 4a,b. Since the e_g band is σ -anti-bonding, the effect is larger for the e_{σ} band, effectively reducing the crystal field splitting. Due to the reduced overlap between the iron and oxygen orbitals, the VB becomes more dominated by the oxygen 2p character, and the conduction band becomes more dominated by the iron 3d character. This implies that the charge density increases at the oxygen sites and decreases at the iron sites. A decreased charge density at the iron sites tends to shift the iron 2p XAS to higher energies. This is indicated with the ϵ value of -0.3 eV. This ϵ -shift suggests that the iron 2p XAS peaks shift less than expected from the 3d levels by 0.3 eV. Conversely, the oxygen 1s XAS shifts to lower energies ($\epsilon = 0.3 \text{ eV}$), indicating that the shifts for the oxygen 1s XAS are increased by 0.3 eV. We conclude that the reduced overlap between the oxygen 2*p* and iron 3*d* orbitals can explain all the experimental observations: 1) the XAS shifts to lower energies, 2) the decrease in the crystal field splitting, and 3) the relatively larger shifts for the oxygen 1s XAS.

3. Conclusion

In this study, we employed the fs-XAS technique at PAL-XFEL to unravel light-driven ultrafast electronic and structural changes in epitaxial LaFeO₃ thin films. We demonstrated that the holes in LaFeO₃ persist twice as long as those in Fe₂O₃, highlighting its potential as a promising candidate for oxygen evolution reaction. Moreover, the CFS in LaFeO₃ is inextricably intertwined with the local charge density of the iron ions. Variations in the CFS can be related to changes in the charge density due to photoexcitation and relaxation effects. The intermediate CFS values of 1.5 eV at 0.2 ps and 1.75 eV at 1 ps obtained from the multiplet calculations describe these transient states captured by our fs-XAS experiments. The photoexcitation initially induces high- and intermediate-spin Fe²⁺ states through LMCT, followed by polaron formation at longer delay times. Upon optical excitation, an electron is moved from the oxygen 2p VB to the iron 3d band, effectively reducing the orbital overlap between iron and oxygen. This leads to a downward energy shift of both the t_{2g} and e_g bands relative to the ground state. These findings shed light on the dynamic coupling between local charge density changes and the CFS in LaFeO₃. In particular, they provide valuable insights into the ultrafast processes governing charge transfer and polaron formation-key mechanisms for understanding the photoexcited behavior of related materials.

4. Experimental Section

fs-XAS Experiments: The fs-XAS spectra were acquired at the Soft Xray Scattering and Spectroscopy (SSS) beamline at the Pohang Accelerator Laboratory X-ray Free Electron Laser (PAL-XFEL) in South Korea. All measurements were performed at room temperature and in the ultra-high

www.advmat.de





vacuum condition of 10^{-8} mbar. The resolution of the experiment was 120 fs, a similar value as in our previous experiments.^[4,44,45] The average number of acquisitions varied between 5 and 10 depending on the signal/noise ratio. The static XAS was measured each time to monitor for potential beam damage. The monochromatized X-rays were normalized using an I₀ monitor and focused onto the sample with Kirkpatrick–Baez mirrors (Figure 1a). The measurements were conducted in electron yield (EY) mode, detected by a micro-channel plate (MCP) detector. The relative timing between the pump and probe was controlled via an optical delay line, and the spatial and temporal overlap (delay = 0) was aligned by observing fluorescence from a thin cerium-doped yttrium aluminum garnet (Ce:YAG) crystal.

Optical Pump: A Ti:sapphire laser with a wavelength of 400 nm (3.10 eV) and a pulse duration of 80 fs was focused on a spot with a full width at half maximum (FWHM) of 100 μ m. The fluence of the pump laser was 13.7 mJ cm⁻². Across the fluence range of 8–30 mJ cm⁻², no significant changes were observed in the transient features or lifetime behavior. At fluences below 3.4 mJ cm⁻², the signal induced in the transients was too weak to provide a sufficient signal-to-noise ratio. The fluences up to 30 mJ cm⁻² did not cause any damage to the sample.

Pulsed Laser Deposition (PLD): The epitaxial LaFeO₃ thin film was grown using pulsed laser deposition (PLD) with in situ RHEED at the MESA+ Institute for Nanotechnology of the University of Twente, in the Netherlands. The Nb:SrTiO₃ (100) substrates purchased from CrysTec GmbH were used for the growth. The Nb:SrTiO₃ substrate was TiO₂ terminated, which has already been demonstrated to suppress the possible extraction of photocarriers generated in Nb:SrTiO₃. [46,47] The O₂ pressure during the growth was 0.01 mbar, the growth temperature was 700 °C, the fluence was 1.9 J cm⁻², the target-substrate distance was 5 cm, and the frequency was 2 Hz. The thickness of the sample was determined to be \approx 30 nm (Figure S9).

Reflection High-Energy Electron Diffraction (RHEED) and Atomic Force Microscopy (AFM): The growth was monitored in situ using reflection high-energy electron diffraction (RHEED), as illustrated in Figure S8a–c (Supporting Information). The topography of the grown films was characterized by atomic force microscopy (AFM) using a Veeco Dimension Icon AFM in tapping mode in the air. The oscillating cantilever is a Tespa-V2 cantilever (Bruker, Netherlands) with a silicon tip with a nominal radius of 20 nm. The AFM images were obtained using the Nanoscope software and treated using the Gwyddion software,^[48] shown in Figure S8d (Supporting Information).

High-Resolution X-Ray Diffraction (HRXRD): The HRXRD characterization was carried out by a Bruker D8 Discover diffractometer with a high-brilliance microfocus Cu rotating anode generator (2.5 kW), hybrid Montel optics, a two bounds monochromator, a 1 mm diameter circular pinhole beam collimator, and an EIGER2 R 500K hybrid photon counting area detector. The detector was operated in 0D mode with a region of interest of 13 × 65 pixels (75 × 75 μ m² each) for the 2 θ - ω scan (Figure S9a). The reciprocal space map (RSM) was obtained with the detector in 1D mode while performing a rocking curve (Figure S9b).

Density Functional Theory (DFT) Calculations: The DFT calculations were performed using the QuantumESPRESSO package.^[49-52] The wavefunctions were expanded with plane waves up to an energy cutoff of 80 Ry and a 6×4×6 Monkhorst-Pack grid was used to sample the Brillouin Zone. PBE-GGA functional was used to describe the exchange-correlation potential, and correlations for Fe *d* and La *f* orbitals were corrected using Hubbard parameters as described in ref. [29]. X-ray absorption spectra were calculated using the XSpectra code (Figure S2).^[52–54]

Multiplet Calculations: The multiplet calculations were carried out using CTM4XAS^[55] and Quanty.^[56–58]

Supporting Information

Supporting Information is available from the Wiley Online Library or from the author.

Acknowledgements

M.L. and F.M.F.d.G received funding from the European Union's Horizon 2020 research and innovation program under the Marie Skłodowska-Curie grant agreement No. 860553. The experiments were performed using the SSS instrument at PAL-XFEL (Proposal No. 2023-2nd-SSS-1002), funded by the Ministry of Science and ICT of Korea. This work was supported by the National Research Foundation of Korea (NRF) grant funded by the Korean government (MSIT) (RS-2024-00355581 and RS-2023-00208787). U.B. acknowledges partial support of this research by NSF through the University of Wisconsin Materials Research Science and Engineering Center (DMR-2309000). This work used the Dutch National e-Infrastructure with the support of the SURF Cooperative using grant no. EINF-7019. M.L. is grateful to Dominic Post, Daniel Monteiro Cunha, and Frank Roesthuis for their support in the lab at the University of Twente. M.L. is grateful to Sharif Bayoumy and Yosua Adisapta Pranata Andoko at the XR Zone of the Delft University of Technology for their guidance on Blender software. The authors appreciate fruitful discussions with Yohei Uemura at EuXFEL.

Conflict of Interest

The authors declare no conflict of interest.

Author Contributions

M.L., U.B., and F.M.F.d.G. conceived the experiments. S.H.P., A.K., and S.K. designed and carried out the fs-XAS experiments. M.L. and S.H.P. analyzed the fs-XAS data. F.J.M., H.J.F.A.B., M.L., and F.M.F.d.G. performed the multiplet calculations and interpreted the fs-XAS data. A.B.M. conducted the DFT calculations. A.B.M., M.L., and F.M.F.d.G. interpreted the DFT calculations. M.L., with the assistance of E.v.d.M., Y.A.B., and I. C. G. v.d.B. prepared the thin film samples using PLD. Y.A.B. and E.M.K. performed the HRXRD and RSM data. C.B. and G.K. supervised the sample preparation. M.L. and F.M.F.d.G. wrote the manuscript with contributions from all the co-authors.

Data Availability Statement

The data that support the findings of this study are openly available in DataverseNL in ref. [59].

Keywords

density funcational theory, femtosecond X-ray absorption spectroscopy, LaFeO₃, multiplet calculations, X-ray free-electron laser

Received: February 12, 2025 Revised: March 30, 2025 Published online:

- S. Nishioka, F. E. Osterloh, X. Wang, T. E. Mallouk, K. Maeda, Nat. Rev. Methods Prim. 2023, 3, 42.
- [2] J. A. Dias, M. A. S. Andrade, H. L. S. Santos, M. R. Morelli, L. H. Mascaro, ChemElectroChem 2020, 7, 3173.
- [3] A. Britz, A. R. Attar, X. Zhang, H.-T. Chang, C. Nyby, A. Krishnamoorthy, S. H. Park, S. Kwon, M. Kim, D. Nordlund, S. Sainio, T. F. Heinz, S. R. Leone, A. M. Lindenberg, A. Nakano, P. Ajayan, P. Vashishta, D. Fritz, M.-F. Lin, U. Bergmann, *Struct. Dyn.* 2021, *8*, 014501.

ADVANCED SCIENCE NEWS

www.advancedsciencenews.com



www.advmat.de

- [4] S. H. Park, A. Katoch, K. H. Chae, S. Gautam, P. Miedema, S. W. Cho, M. Kim, R.-P. Wang, M. Lazemi, F. de Groot, S. Kwon, *Nat. Commun.* 2022, 13, 2531.
- [5] T. Lojewski, M. F. Elhanoty, L. Le Guyader, O. Grånäs, N. Agarwal, C. Boeglin, R. Carley, A. Castoldi, C. David, C. Deiter, F. Döring, R. Engel, F. Erdinger, H. Fangohr, C. Fiorini, P. Fischer, N. Gerasimova, R. Gort, F. deGroot, K. Hansen, S. Hauf, D. Hickin, M. Izquierdo, B. E. Van Kuiken, Y. Kvashnin, C.-H. Lambert, D. Lomidze, S. Maffessanti, et al., *Mater. Res. Lett.* **2023**, *11*, 655.
- [6] D. Pile, Nat. Photon. 2024, 18, 640.
- [7] P. Emma, R. Akre, J. Arthur, R. Bionta, C. Bostedt, J. Bozek, A. Brachmann, P. Bucksbaum, R. Coffee, F.-J. Decker, Y. Ding, D. Dowell, S. Edstrom, A. Fisher, J. Frisch, S. Gilevich, J. Hastings, G. Hays, P. Hering, Z. Huang, R. Iverson, H. Loos, M. Messerschmidt, A. Miahnahri, S. Moeller, H.-D. Nuhn, G. Pile, D. Ratner, J. Rzepiela, D. Schultz, et al., *Nat. Photon* **2010**, *4*, 641.
- [8] B. W. J. McNeil, N. R. Thompson, Nat. Photon 2010, 4, 814.
- [9] L. Mercadier, A. Benediktovitch, S. Krusic, J. J. Kas, J. Schlappa, M. Agåker, R. Carley, G. Fazio, N. Gerasimova, Y. Y. Kim, L. Le Guyader, G. Mercurio, S. Parchenko, J. J. Rehr, J.-E. Rubensson, S. Serkez, M. Stransky, M. Teichmann, Z. Yin, M. Zitnik, A. Scherz, B. Ziaja, N. Rohringer, *Nat. Phys.* **2024**, *20*, 1564.
- [10] X-Ray Free Electron Lasers: Applications in Materials, Chemistry and Biology, (Eds.: U. Bergmann, V. Yachandra, J. Yano), Royal Society of Chemistry, Cambridge 2017.
- [11] A. S. M. Ismail, Y. Uemura, S. H. Park, S. Kwon, M. Kim, H. Elnaggar, F. Frati, Y. Niwa, H. Wadati, Y. Hirata, Y. Zhang, K. Yamagami, S. Yamamoto, I. Matsuda, U. Halisdemir, G. Koster, B. M. Weckhuysen, F. M. F. de Groot, *Phys. Chem. Chem. Phys.* **2020**, *22*, 2685.
- [12] Y. Uemura, A. S. M. Ismail, S. H. Park, S. Kwon, M. Kim, H. Elnaggar, F. Frati, H. Wadati, Y. Hirata, Y. Zhang, K. Yamagami, S. Yamamoto, I. Matsuda, U. Halisdemir, G. Koster, C. Milne, M. Ammann, B. M. Weckhuysen, F. M. F. de Groot, J. Phys. Chem. Lett. 2022, 13, 4207.
- [13] Y. Uemura, A. S. M. Ismail, S. H. Park, S. Kwon, M. Kim, Y. Niwa, H. Wadati, H. Elnaggar, F. Frati, T. Haarman, N. Höppel, N. Huse, Y. Hirata, Y. Zhang, K. Yamagami, S. Yamamoto, I. Matsuda, T. Katayama, T. Togashi, S. Owada, M. Yabashi, U. Halisdemir, G. Koster, T. Yokoyama, B. M. Weckhuysen, F. M. F. de Groot, *J. Phys. Chem. C* **2021**, *125*, 7329.
- [14] J. Vura-Weis, C.-M. Jiang, C. Liu, H. Gao, J. M. Lucas, F. M. F. de Groot, P. Yang, A. P. Alivisatos, S. R. Leone, J. Phys. Chem. Lett. 2013, 4, 3667.
- [15] K. J. May, D. P. Fenning, T. Ming, W. T. Hong, D. Lee, K. A. Stoerzinger, M. D. Biegalski, A. M. Kolpak, Y. Shao-Horn, J. Phys. Chem. Lett. 2015, 6, 977.
- [16] J. Lüning, F. Nolting, A. Scholl, H. Ohldag, J. W. Seo, J. Fompeyrine, J.-P. Locquet, J. Stöhr, *Phys. Rev. B* 2003, 67, 214433.
- [17] E. Folven, T. Tybell, A. Scholl, A. Young, S. T. Retterer, Y. Takamura, J. K. Grepstad, Nano Lett. 2010, 10, 4578.
- [18] Q. Che, I. C. G. van den Bosch, P. T. P. Le, M. Lazemi, E. van der Minne, Y. A. Birkhölzer, M. Nunnenkamp, M. L. J. Peerlings, O. V. Safonova, M. Nachtegaal, G. Koster, C. Baeumer, P. de Jongh, F. M. F. de Groot, J. Phys. Chem. C 2024, 128, 5515.
- [19] A. Scholl, J. Stohr, J. Luning, J. W. Seo, J. Fompeyrine, H. Siegwart, J.-P. Locquet, F. Nolting, S. Anders, E. E. Fullerton, M. R. Scheinfein, H. A. Padmore, *Science* **2000**, *287*, 1014.
- [20] I. Ko, H.-S. Kang, H. Heo, C. Kim, G. Kim, C.-K. Min, H. Yang, S. Baek, H.-J. Choi, G. Mun, B. Park, Y. Suh, D. Shin, J. Hu, J. Hong, S. Jung, S.-H. Kim, K. Kim, D. Na, S. Park, Y. Park, Y. Jung, S. Jeong, H. Lee, S. Lee, S. Lee, B. Oh, H. Suh, J.-H. Han, M. Kim, et al., *Appl. Sci.* 2017, 7, 479.
- [21] M. D. Scafetta, A. M. Cordi, J. M. Rondinelli, S. J. May, J. Phys.: Condens. Matter 2014, 26, 505502.
- [22] F. de Groot, A. Kotani, Core Level Spectroscopy of Solids, CRC Press, Boca Raton, FL 2008.

- [23] F. Frati, M. O. J. Y. Hunault, F. M. F. de Groot, Chem. Rev. 2020, 120, 4056.
- [24] M. Abbate, F. M. F. de Groot, J. C. Fuggle, A. Fujimori, O. Strebel, F. Lopez, M. Domke, G. Kaindl, G. A. Sawatzky, M. Takano, Y. Takeda, H. Eisaki, S. Uchida, *Phys. Rev. B* **1992**, *46*, 4511.
- [25] D. D. Sarma, O. Rader, T. Kachel, A. Chainani, M. Mathew, K. Holldack, W. Gudat, W. Eberhardt, *Phys. Rev. B* 1994, 49, 14238.
- [26] A. Chainani, M. Mathew, D. D. Sarma, Phys. Rev. B 1993, 48, 14818.
- [27] J. Suntivich, W. T. Hong, Y.-L. Lee, J. M. Rondinelli, W. Yang, J. B. Goodenough, B. Dabrowski, J. W. Freeland, Y. Shao-Horn, J. Phys. Chem. C 2014, 118, 1856.
- [28] S. Lafuerza, G. Subías, J. García, S. Di Matteo, J. Blasco, V. Cuartero, C. R. Natoli, J. Phys.: Condens. Matter 2011, 23, 325601.
- [29] I. Timrov, P. Agrawal, X. Zhang, S. Erat, R. Liu, A. Braun, M. Cococcioni, M. Calandra, N. Marzari, D. Passerone, *Phys. Rev. Res.* 2020, 2, 033265.
- [30] F. M. F. de Groot, M. Grioni, J. C. Fuggle, J. Ghijsen, G. A. Sawatzky, H. Petersen, *Phys. Rev. B* **1989**, 40, 5715.
- [31] P. Kuiper, B. G. Searle, P. Rudolf, L. H. Tjeng, C. T. Chen, Phys. Rev. Lett. 1993, 70, 1549.
- [32] A. Katoch, S. H. Park, K. Jeong, M. Lazemi, R.-P. Wang, H. S. Ahn, T. K. Kim, F. M. F. de Groot, S. Kwon, *Adv. Opt. Mater.* 2024, *12*, 2401386.
- [33] F. M. F. de Groot, H. Elnaggar, F. Frati, R.-P. Wang, M. U. Delgado-Jaime, M. van Veenendaal, J. Fernandez-Rodriguez, M. W. Haverkort, R. J. Green, G. van der Laan, Y. Kvashnin, A. Hariki, H. Ikeno, H. Ramanantoanina, C. Daul, B. Delley, M. Odelius, M. Lundberg, O. Kuhn, S. I. Bokarev, E. Shirley, J. Vinson, K. Gilmore, M. Stener, G. Fronzoni, P. Decleva, P. Kruger, M. Retegan, Y. Joly, C. Vorwerk, et al., J. Electron Spectrosc. Relat. Phenom. **2021**, 249, 147061.
- [34] F. de Groot, Coord. Chem. Rev. 2005, 249, 31.
- [35] F. M. F. de Groot, J. C. Fuggle, B. T. Thole, G. A. Sawatzky, *Phys. Rev.* B 1990, 42, 5459.
- [36] F. M. F. de Groot, J. C. Fuggle, B. T. Thole, G. A. Sawatzky, *Phys. Rev.* B 1990, 41, 928.
- [37] C. Franchini, M. Reticcioli, M. Setvin, U. Diebold, Nat. Rev. Mater. 2021, 6, 560.
- [38] L. M. Carneiro, S. K. Cushing, C. Liu, Y. Su, P. Yang, A. P. Alivisatos, S. R. Leone, *Nat. Mater.* 2017, *16*, 819.
- [39] B. K. Ridley, Quantum Processes in Semiconductors, Oxford University Press, Oxford 2013.
- [40] G. L. Stamokostas, G. A. Fiete, Phys. Rev. B 2018, 97, 085150.
- [41] Y. Tokura, N. Nagaosa, Science 2000, 288, 462.
- [42] D. I. Khomskii, Transition Metal Compounds, Cambridge University Press, Cambridge 2014.
- [43] S. P. Cramer, F. M. F. DeGroot, Y. Ma, C. T. Chen, F. Sette, C. A. Kipke, D. M. Eichhorn, M. K. Chan, W. H. Armstrong, E. Libby, G. Christou, S. Brooker, V. McKee, O. C. Mullins, J. C. Fuggle, *J. Am. Chem. Soc.* **1991**, *113*, 7937.
- [44] S. H. Park, M. Kim, C.-K. Min, I. Eom, I. Nam, H.-S. Lee, H.-S. Kang, H.-D. Kim, H. Y. Jang, S. Kim, S.-M. Hwang, G.-S. Park, J. Park, T.-Y. Koo, S. Kwon, *Rev. Sci. Instrum.* **2018**, *89*, 055105.
- [45] S. H. Park, J. Yoon, C. Kim, C. Hwang, D.-H. Kim, S.-H. Lee, S. Kwon, J. Synchrotron Rad. 2019, 26, 1031.
- [46] K. Nakamura, H. Mashiko, K. Yoshimatsu, A. Ohtomo, Appl. Phys. Lett. 2016, 108, 211605.
- [47] G. Koster, B. L. Kropman, G. J. H. M. Rijnders, D. H. A. Blank, H. Rogalla, Appl. Phys. Lett. 1998, 73, 2920.
- [48] D. Nečas, P. Klapetek, Open Phys. 2012, 10, 181.
- [49] P. Giannozzi, S. Baroni, N. Bonini, M. Calandra, R. Car, C. Cavazzoni, D. Ceresoli, G. L. Chiarotti, M. Cococcioni, I. Dabo, A. Dal Corso, S. de Gironcoli, S. Fabris, G. Fratesi, R. Gebauer, U. Gerstmann,

SCIENCE NEWS _____



www.advmat.de

C. Gougoussis, A. Kokalj, M. Lazzeri, L. Martin-Samos, N. Marzari, F. Mauri, R. Mazzarello, S. Paolini, A. Pasquarello, L. Paulatto, C. Sbraccia, S. Scandolo, G. Sclauzero, A. P. Seitsonen, et al., *J. Phys.: Condens. Matter* **2009**, *21*, 395502.

- [50] P. Giannozzi, O. Andreussi, T. Brumme, O. Bunau, M. Buongiorno Nardelli, M. Calandra, R. Car, C. Cavazzoni, D. Ceresoli, M. Cococcioni, N. Colonna, I. Carnimeo, A. Dal Corso, S. de Gironcoli, P. Delugas, R. A. DiStasio, A. Ferretti, A. Floris, G. Fratesi, G. Fugallo, R. Gebauer, U. Gerstmann, F. Giustino, T. Gorni, J. Jia, M. Kawamura, H.-Y. Ko, A. Kokalj, E. Küçükbenli, M. Lazzeri, et al., J. Phys.: Condens. Matter 2017, 29, 465901.
- [51] M. Taillefumier, D. Cabaret, A.-M. Flank, F. Mauri, Phys. Rev. B 2002, 66, 195107.
- [52] C. Gougoussis, M. Calandra, A. P. Seitsonen, F. Mauri, Phys. Rev. B 2009, 80, 075102.

- [53] D. Cabaret, M. Calandra, in XSpectra : A Density-Functional-Theory-Based Plane-Wave Pseudopotential Code for XANES Calculation, Wiley, Hoboken, NJ 2020, pp. 851–856.
- [54] O. Bunău, M. Calandra, Phys. Rev. B 2013, 87, 205105.
- [55] E. Stavitski, F. M. F. de Groot, *Micron* **2010**, *41*, 687.
- [56] Y. Lu, M. Höppner, O. Gunnarsson, M. W. Haverkort, Phys. Rev. B 2014, 90, 085102.
- [57] M. W. Haverkort, G. Sangiovanni, P. Hansmann, A. Toschi, Y. Lu, S. Macke, *EPL* 2014, 108, 57004.
- [58] M. W. Haverkort, M. Zwierzycki, O. K. Andersen, Phys. Rev. B 2012, 85, 165113.
- [59] M. Lazemi, F. M. F. de Groot, Ultrafast light-driven electronic and structural changes in LaFeO₃ perovskites probed by femtosecond X-ray absorption spectroscopy, DataverseNL, V1, 2025, https://doi.org/10.34894/GYDZZW.