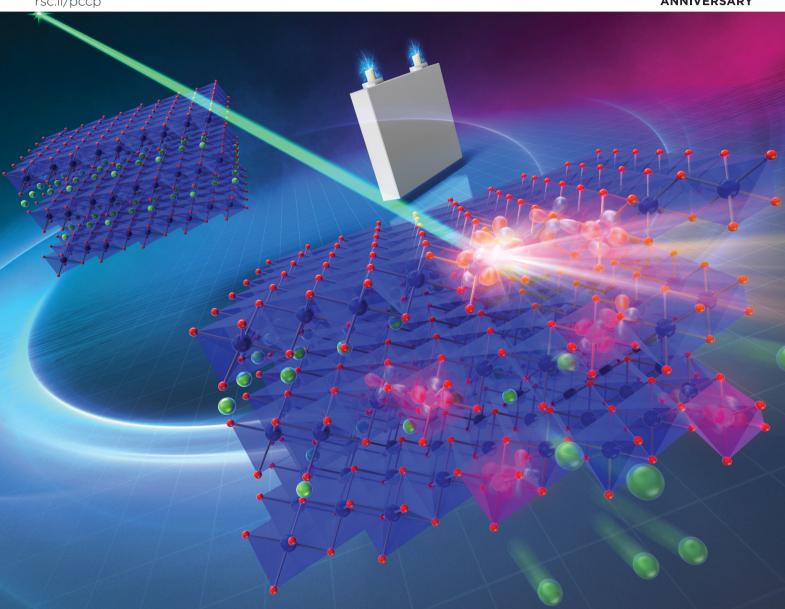
Volume 27 Number 8 28 February 2025 Pages 4009-4498

# PCCP

Physical Chemistry Chemical Physics

rsc.li/pccp





ISSN 1463-9076



#### **PAPER**

Daisuke Asakura et al. Elucidation of the  $\mathrm{Co^{4+}}$  state with strong charge-transfer effects in charged  $\mathrm{LiCoO_2}$  by resonant soft X-ray emission spectroscopy at the Co  $L_3$  edge



# PCCP



**View Article Online PAPER** 



Cite this: Phys. Chem. Chem. Phys., 2025, 27, 4092

# Elucidation of the Co4+ state with strong charge-transfer effects in charged LiCoO<sub>2</sub> by resonant soft X-ray emission spectroscopy at the Co L<sub>3</sub> edge

Daisuke Asakura, (10 \*abc Takaaki Sudayama, a Yusuke Nanba, (10 †a Eiji Hosono, (10 abc Hisao Kiuchi, de Kosuke Yamazoe, de Jun Miyawaki, de Yoshihisa Harada, code Atsuo Yamada, pfg Ru-Pan Wang sh and Frank M. F. de Groot h

To understand the electronic-structure change of LiCoO2, a widely used cathode material in Li-ion batteries, during charge-discharge, we performed ex situ soft X-ray absorption spectroscopy (XAS) and resonant soft X-ray emission spectroscopy (RXES) of the Co L<sub>3</sub> edge in combination with chargetransfer multiplet calculations. The RXES profile significantly changed for the charged state at 4.2 V vs. Li/Li<sup>+</sup>, corresponding to the oxidation reaction from a Co<sup>3+</sup> low-spin state for the initial state, while the XAS profile exhibited small changes. For the 4.2-V charged state, we confirmed that approximately half of the initial Co<sup>3+</sup> ions were oxidized to Co<sup>4+</sup> ions. The multiplet calculation of the RXES results revealed that the  $Co^{4+}$  state has a negative charge-transfer energy and the  $d^6 \underline{L}$  state ( $\underline{L}$  is a ligand hole) is the most stable. Therefore, the O 2p hole created by the strong charge-transfer effect plays a major role in the redox reaction of LiCoO<sub>2</sub>.

Received 30th September 2024, Accepted 7th December 2024

DOI: 10.1039/d4cp03759f

rsc.li/pccp

#### 1. Introduction

In the past few decades, Li-ion batteries (LIBs) have been extensively developed for various applications. To apply LIBs to electric vehicles and large-scale energy-storage systems, their energy density and safety need to be improved. Increasing the charge-discharge capacity of the cathodes of LIBs is particularly important for increasing the energy density. The chargedischarge capacity is directly connected with the redox reaction. In the case of LiCoO2, which is a typical cathode material, the formal valence of Co changes according to the equation  $LiCo^{3+}O_2 \Leftrightarrow 0.5Li^+ + 0.5e^- + Li_{0.5}Co^{4+}_{0.5}Co^{3+}_{0.5}O_2$  with charge (Li extraction)/discharge (Li insertion) within a cutoff voltage range of 3.0-4.2 V (vs. Li/Li<sup>+</sup>), where the capacity is approximately 150 mA h g<sup>-1</sup>.<sup>1,2</sup> To confirm the Co<sup>3+</sup>/Co<sup>4+</sup> redox reaction, many groups have investigated the electronic structure of LiCoO<sub>2</sub> by various methods.<sup>1-9</sup> X-ray absorption spectroscopy (XAS) is one of the most powerful tools to analyze the electronic structure of cathode materials. In the case of 3d transitionmetal (TM) oxides (TMOs) involving LiCoO2, TM K-edge (mainly the 1s  $\rightarrow$  4p transition) and L<sub>2,3</sub>-edge XAS (2p<sub>1/2,3/2</sub>  $\rightarrow$  3d transition) have been extensively performed. For example, the Co<sup>3+</sup> low-spin state has been confirmed to be the initial state of LiCoO<sub>2</sub> by Co K-edge and L<sub>2,3</sub>-edge XAS.<sup>8-16</sup> For the charge process of LiCoO2, Co K-edge XAS studies have shown that the main peak shifts to the high-energy side, and Co L2,3-edge XAS studies have revealed that the XAS line shape changes, which suggest the oxidation reaction from the Co3+ state to the Co4+ state.15,16

In this study, we focus on the Co4+ state in the charge process because the electronic configuration and Co 3d-O 2p

<sup>&</sup>lt;sup>a</sup> Research Institute for Energy Conservation, National Institute of Advanced Industrial Science and Technology (AIST), Tsukuba, Ibaraki 305-8585, Japan. E-mail: daisuke-asakura@aist.go.jp

<sup>&</sup>lt;sup>b</sup> Global Zero Emission Research Center, National Institute of Advanced Industrial Science and Technology (AIST), Tsukuba, Ibaraki 305-8589, Japan

<sup>&</sup>lt;sup>c</sup>AIST-UTokyo Advanced Operando Measurement Technology Open Innovation Laboratory (Operando OIL), Kashiwa, Chiba 277-8565, Japan

<sup>&</sup>lt;sup>d</sup> Institute for Solid State Physics (ISSP), The University of Tokyo, Kashiwa, Chiba 277-8581, Iapan

<sup>&</sup>lt;sup>e</sup> Synchrotron Radiation Research Organization (SRRO), The University of Tokyo, Sendai, Miyagi 980-8572, Japan

f Department of Chemical System Engineering, The University of Tokyo, Bunkyo-ku, Tokyo 113-8656, Japan

g Sungkyunkwan University Institute of Energy Science & Technology (SIEST), Sungkyunkwan University, Suwon 16419, South Korea

<sup>&</sup>lt;sup>h</sup> Debye Institute for Nanomaterials Science, Utrecht University, 3584 GG Utrecht, The Netherlands

<sup>†</sup> Present address: Research Initiative for Supra-Materials, Shinshu University, Nagano, Nagano 380-8553, Japan.

<sup>‡</sup> Present address: NanoTerasu Center, National Institutes for Quantum Science and Technology, Sendai, Miyagi 980-8579, Japan.

<sup>§</sup> Present address: Deutsches Elektronen Synchrotron DESY, Hamburg 22607, Germany.

orbital hybridization of the Co-O bond have been unclear. Resonant soft X-ray emission spectroscopy (RXES) combined with theoretical charge-transfer (CT) multiplet calculations is a promising method to investigate the Co 3d electronic structure. RXES is the secondary process after XAS, reflecting the element and orbital-selective occupied (valence) state. 17-19 Recently, RXES has attracted considerable interest for studying electrode materials, 20-22 including Li-rich layered oxides with oxygen redox. 23-29 For LiCoO2, the oxygen redox reaction has been investigated by O K RXES, 30,31 but the reaction occurs at voltage around 4.6 V vs. Li/Li<sup>+</sup>, which is higher than the voltage range in which the Co3+/Co4+ redox occurs in practical applications (below  $\sim 4.2 \text{ V } \text{ vs. Li/Li}^+$ ). Here, we report a Co L<sub>3</sub> (2p-3d-2p resonance) ex situ RXES study of LiCoO2 in three states (the initial state, the 4.2-V charged state, and the discharged state after 4.2-V charging) to clarify the intrinsic role of the O 2p-Co 3d CT effect in the redox reaction below 4.2 V.

#### 2. Method

LiCoO<sub>2</sub> powder was fabricated by following the procedure below. First, CoCl<sub>2</sub>·6H<sub>2</sub>O (0.0025 mol) and LiNO<sub>3</sub> (0.020 mol) were dissolved in H<sub>2</sub>O (10 ml). The solution was then dried at 140 °C in vacuum for 17 h. The dried reactant was placed in a Al<sub>2</sub>O<sub>3</sub> crucible with a lid. The crucible was placed in a muffle furnace and heated at 800 °C in air for 3 h. Finally, the sample was washed with deionized water and dried under vacuum conditions. The LiCoO<sub>2</sub> powder was pasted with super-P Li as a carbon additive to increase the conductivity and polytetrafluoroethylene as a binder. The paste was then pressed onto Al mesh as a current collector to fabricate the working electrode. A three-electrode beaker cell was assembled with the LiCoO<sub>2</sub> working electrode, a Li-metal counter electrode, a Li-metal reference electrode, and a 1 M LiClO<sub>4</sub>/ethylene carbonatediethyl carbonate electrolyte solution. The charge/discharge experiments of the beaker cells were performed by cyclic voltammetry (CV). The scan speed was set to 0.5 mV s<sup>-1</sup>, and the cutoff voltages were 4.2 V (vs. Li/Li<sup>+</sup>) for charging and 3.0 V for discharging. The charged/discharged LiCoO<sub>2</sub> samples were removed from the cell in a glove box, and they were then transferred to a vacuum chamber using a transfer vessel without exposure to air.

The XAS measurements were performed at BL07LSU of SPring-8<sup>32</sup> using the surface-sensitive total-electron yield (TEY) mode. The RXES experiments were performed using the ultrahigh-resolution X-ray emission spectrometer HORNET at BL07LSU of SPring-8.33 The energy resolution of the incident beam was 100 meV, and the total energy resolution for the RXES measurements was  $\Delta E$  = 330 meV at 785.0 eV. All of the XAS and RXES measurements were performed at room temperature. We also performed spectral analyses using Quanty software programs based on CT multiplet calculation. 34,35 In the calculation, the Co 3d orbitals with the Co 2p core-hole effect and O 2p orbitals in a CoO6 octahedron were considered. To describe the ligand-to-metal CT, configuration interactions were used as a linear combination of the basis of each electron configuration with the CT effect  $(d^n, d^{n+1}\underline{L}, and$  $d^{n+2}\underline{L}^2$ , where  $\underline{L}$  denotes a ligand hole).

#### Results and discussion

The CV curves of LiCoO<sub>2</sub> are shown in Fig. 1. The voltage range was set to 3.0-4.2 V (vs. Li/Li<sup>+</sup>). The open-circuit voltage before cycling was 3.2 V. A broad structure was observed around 4.1 V in the charge process. This indicates the oxidation reaction due to Li extraction from LiCoO<sub>2</sub>. For the discharge process, the reduction peak was observed at 3.8 V owing to reinsertion of Li.

To investigate the unoccupied Co 3d states and select the incident photon energy (excitation energy) for RXES, TEY XAS measurements were performed. The Co L<sub>3</sub>-edge TEY XAS results for LiCoO2 are shown in Fig. 2. According to the XAS line profiles in previous reports, 8,9,14 the initial and discharged states are attributed to the Co<sup>3+</sup> state, suggesting a reversible redox reaction for the Co 3d state. The small multiplet splitting for the initial and discharged states suggests that the eg orbital is almost unoccupied in the Co3+ (d6) low-spin (LS) state.14 The LS  $Co^{3+}$  has occupied six  $t_{2g}$  orbitals (which should be further split into e'g and a1g orbitals in the lower symmetry as reported in ref. 14) and all the  $e_g$  orbitals are empty.

The main peak position in the 4.2-V charged-state spectrum slightly shifted to the higher energy region and the shoulder structures at 776.8-778.4 and 781.0-782.0 eV were enhanced (indicated by the arrows in Fig. 2). These results are similar to those in previous reports. 8-14 The changes for the 4.2-V charged state indicate the appearance of the Co4+ state. The slight shift of the main peak suggests oxidation of Co, which should be accompanied with the slight increase of the structure at 781.0-782.0 eV. The increase of the shoulder structure at 776.8-778.4 eV is attributed to  $t_{2g}$  hole (on the  $a_{1g}$  orbital in the lower symmetry) created by charge.14 Moreover, it is expected that approximately half of the Co<sup>3+</sup> sites remained (i.e., approximately 0.5Li was extracted) at the upper cutoff voltage of 4.2 V,

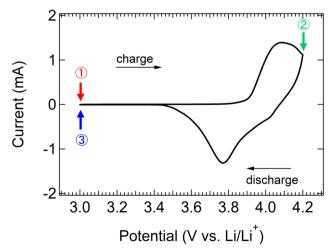


Fig. 1 CV curves of LiCoO<sub>2</sub> with a scan speed of 0.5 mV s<sup>-1</sup>.

**PCCP** 

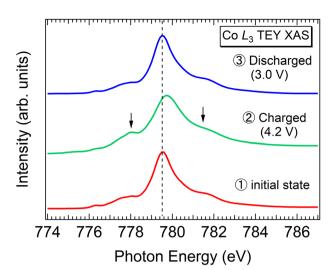


Fig. 2 Ex situ Co L<sub>3</sub>-edge TEY XAS spectra corresponding to the three states in Fig. 1. The dotted line indicates the excitation energy (779.5 eV) for the RXES measurements

which is usual value for practical use to maintain the host framework.1,2

To further investigate the Co<sup>3+</sup> state in the initial state and determine how it changes by the charge process, the excitation energy for RXES was selected to be 779.5 eV, which is the Co L<sub>3</sub>edge main peak position for the initial and discharged states (indicated by the dashed line in Fig. 2). The same excitation energy was used for the charged sample.

The RXES results at the Co L<sub>3</sub> edge are shown in Fig. 3. The peak at 779.5 eV corresponds to the elastic scattering where the emission energy equals the excitation energy. The sharp peaks from 775.0 to 779.0 eV for the initial state are related to the dd excitations. The broad structure below 775.0 eV is mainly related to the CT excitation between the Co 3d and O 2p orbitals. In addition, some fluorescence components should be involved for the whole range. These features for the initial state are almost the same as those for the discharged state, suggesting a reversible redox reaction of Co, as for the XAS results. The line shape for the 4.2-V charged state is greatly different from those for the initial and discharged states. Large changes in the Co 3d electronic structure by charging were observed for Co L3 RXES, while Co L3-edge XAS showed small changes. In particular, the three dd-excitation peaks observed in the initial state greatly changed in the 4.2-V charged state. These changes can be attributed to the partial oxidation of the Co<sup>3+</sup> state to the Co<sup>4+</sup> state.

To analyze the RXES spectra, we performed multiplet calculations including CT effects using the Quanty program. The RXES spectrum with 779.5-eV excitation includes a fluorescence component that was not included in the calculation. For the initial state, we also measured the RXES spectrum with 793.8-eV excitation corresponding to the post-edge region of the Co L<sub>3</sub> edge. The RXES spectrum with 793.8-eV excitation (on-resonance for the  $L_2$  edge) below  $\sim 780$  eV in the emissionenergy scale is regarded as the off-resonant spectrum for the

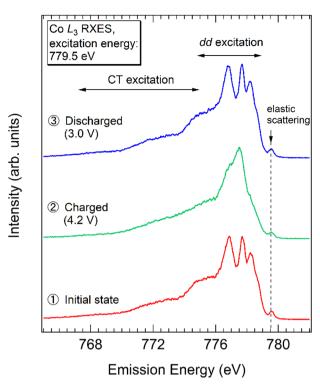


Fig. 3 Ex situ RXES results at the Co L<sub>3</sub> edge of LiCoO<sub>2</sub> for the three states shown in Fig. 1.

L<sub>3</sub> edge. We subtracted the off-resonant spectrum from the onresonant 779.5-eV RXES spectrum to remove the fluorescence. The subtracted spectrum (spectrum A) is compared with the Co<sup>3+</sup> CT multiplet calculation with a CoO<sub>6</sub> cluster model in Fig. 4(a). The calculated result well reproduces the experiment, except for slight differences of the peak intensities of the dd excitations around 778 eV. In addition, the elastic scattering at 779.5 eV in the experimental spectrum is not reproduced well, because it depends on the experimental settings, including the polarization of the incident and emitted photons. The electronic-structure parameters used in the calculation are summarized in Table 1. For the calculation, Oh symmetry was considered for simplicity, while it has been reported that the  $CoO_6$  octahedron has trigonal  $D_{3d}$  symmetry slightly deforming compared to the Oh symmetry.36 The crystal-field splitting 10 Dq value of 1.4 eV is sufficiently strong to maintain the d<sup>6</sup> LS state<sup>37,38</sup> considering the Tanabe-Sugano diagram for this configuration. The CT energy ( $\Delta$ ), corresponding to the energy difference of  $E(d^{7}\underline{L}) - E(d^{6})$  ( $\underline{L}$  is a ligand hole), is 0.6 eV, resulting in a strong CT effect. Indeed, the calculated weight of charge-transferred  $d^{7}\underline{L}$  configuration (52.1%) is higher than that of d<sup>6</sup> configuration (39.0%) as shown in Table 2.

For the 4.2-V charged state, to extract the Co<sup>4+</sup> component, we performed a difference analysis. Firstly, we subtracted the offresonant spectrum for the charged state from the on-resonant 779.5-eV RXES spectrum in the same way for the initial state, which is spectrum B in Fig. 4(b). Next, a difference spectrum (spectrum C) was constructed using the formula of [spectrum B (charged)]  $-0.5 \times$  [spectrum A (initial)] (Fig. 4(c)). Here, we

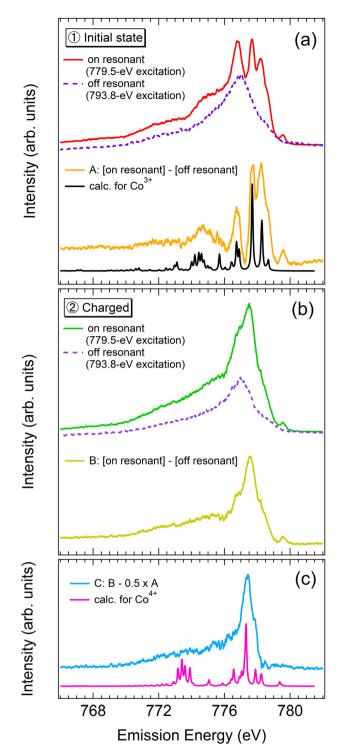


Fig. 4 (a) Subtraction of the off-resonant spectrum from the on resonant spectrum for the initial state and fitting result of the Co3+ state. (b) Subtraction of the off-resonant spectrum from the on resonant spectrum for the charged state. (c) Difference analysis for the charged state and fitting result of the Co<sup>4+</sup> state.

assumed that the Co3+ state was unchanged between the initial and charged states. The coefficient of 0.5 in the formula is an assumption based on the nominal chemical equation of  $LiCo^{3+}O_2 \Leftrightarrow 0.5Li^+ + 0.5e^- + Li_{0.5}Co^{3.5+}O_2$ , while it is difficult to

Table 1 Electronic-structure parameters used in the calculations (in eV)

	$\mathrm{Co}^{3+}\left(O_{\mathrm{h}} \text{ symmetry}\right)$	$\mathrm{Co}^{4+}\left(O_{\mathrm{h}} \text{ symmetry}\right)$
10 Dq	1.4	1.6
Δ	0.6	-1.0
$U_{ m dd}$	6.5	7.0
$U_{ m dd}$ $Q$	7.5	7.5
$V(e_g)$	2.2	3.0
$V(e_g)$ $V(t_{2g})$	1.1	1.5

Table 2 Calculated weights of each electron configuration and calculated effective 3d electron numbers  $\langle n_d \rangle$ 

$Co^{3+}$ ( $O_h$ symmetry)		$\mathrm{Co}^{4+}\left(O_{\mathrm{h}} \text{ symmetry}\right)$	
$egin{array}{c} d^6 \ d^7 \underline{\mathrm{L}} \ d^8 \underline{\mathrm{L}}^2 \ \langle n_\mathrm{d}  angle \end{pmatrix}$	39.0% 52.1% 8.9% 6.7	$egin{array}{c}  ext{d}^5 \  ext{d}^6  ext{L} \  ext{d}^7  ext{L}^2 \  ext{\langle} n_ ext{d} angle \end{array}$	27.3% 57.1% 15.6% 5.9

accurately estimate the ratio of Co3+/Co4+ from the intensity of RXES spectra.

The difference in line shape between spectrum C and spectrum A suggests the oxidation reaction of the Co<sup>3+</sup> LS to Co4+ LS states. The shoulder structure of the 777.9-eV peak in the dd excitation region of spectrum C should correspond to the decrease of the 778.3-eV peak in spectrum A, suggesting the removal of an electron from the fully occupied t2g band of the Co<sup>3+</sup> LS state in the initial state as Fig. 5 shows. Spectrum C was successfully fitted by the Co4+ calculation for the dd and CT excitations (Fig. 4(c)). Indeed, the Co<sup>4+</sup> calculation reproduces the decrease of the 777.9-eV peak. The electronic-structure parameters for the Co4+ calculation are summarized in Table 1. For  $Co^{4+}$ ,  $\Delta$  is defined as  $E(d^6\underline{L}) - E(d^5)$ . The  $\Delta$  value was determined to be -1.0 eV. The negative  $\Delta$  means that the charge-transferred d<sup>6</sup>L configuration is considerably dominant compared to the  $d^5$  configuration. Negative  $\Delta$  in  $Co^{4+}$  configuration has been also found for SrCoO<sub>3</sub>. 39 Indeed, the calculated weight of d<sup>5</sup> configuration is as low as 27.3%, while that of

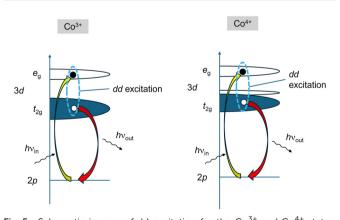


Fig. 5 Schematic images of dd excitation for the Co<sup>3+</sup> and Co<sup>4+</sup> states. For simplicity,  $O_h$  symmetry is considered. The yellow and red allows indicate the absorption and emission processes. The 779.5-eV excitation of RXES in Fig. 3 and 4 corresponds to the excitation (absorption) to only the  $e_q$  band for the  $Co^{4+}$  state.

**PCCP** Paper

charge-transferred d<sup>6</sup>L configuration is calculated to be 57.1% (Table 2). The calculated effective 3d electron number  $\langle n_{\rm d} \rangle$  for Co<sup>4+</sup> is 5.9, which is much larger than 5.0 (Table 2). Moreover, the calculated transfer integral between the Co 3d and O 2p orbitals  $V(e_g)$  is 3.0 eV, which suggests that the hybridization of the  $\sigma$  bonding is not weak, and  $V(t_{2g})$  for the  $\pi$  bonding is 1.5 eV. Therefore, the CT effect of the Co<sup>4+</sup> state is further stronger than that of the Co<sup>3+</sup> state. These results indicate that the O 2p orbital plays an essential role in the redox reaction. The holes created by the charge reaction tend to be located on the p-d hybridized orbital. In that the holes are not localized on the Co 3d orbital, the consideration is partly consistent with previous XAS,8 X-ray Compton scattering,7 and theoretical studies<sup>40</sup> which suggest that the hole should be on the O 2p orbital rather than the Co 3d orbital. We conclude that the O 2p-Co 3d hybridized orbital is redox active for LiCoO<sub>2</sub>.

A similar mechanism has been reported in Mn L-edge XAS and RXES studies of LiMn<sub>2</sub>O<sub>4</sub> with the Mn<sup>3+</sup>/Mn<sup>4+</sup> redox couple<sup>22,41</sup> and Ni L-edge XAS studies of NaFe<sub>0.5</sub>Ni<sub>0.5</sub>O<sub>2</sub> with the Ni<sup>3+</sup>/Ni<sup>4+</sup> redox couple for the Na-ion insertion/extraction.<sup>42</sup> In general, the p-d hybridization in TMOs is particularly important for the physical properties. 43,44 The p-d hybridization in TMOs tends to become stronger for higher oxidation states, such as the tetravalent state. 45 Thus, the redox reaction in TMO-based cathode materials should be considered for the p-d hybridization in the [TM]O<sub>6</sub> octahedron rather than the localized 3d orbital of the TM. The situation is different from that for polyanion-type LiFePO<sub>4</sub>, in which the localized Fe 3d orbital in the FeO<sub>6</sub> octahedron mainly contribute to the redox reaction.<sup>20</sup> LiFePO<sub>4</sub> is also a prototypical cathode material, but it should be separately classified from TMO-based cathode materials such as LiCoO2 and LiMn2O4 in terms of the electronic structure and redox reaction.

The calculated spectra for both the Co3+ and Co4+ states agreed the experimental RXES spectra (Fig. 4), even though the symmetry in the calculation was simplified to Oh symmetry. One reason for the agreement is that the degree of deformation of the CoO<sub>6</sub> octahedron from O<sub>h</sub> symmetry is not large. Another reason is the strong CT effects and strong p-d hybridization. It is most likely that the interaction of the Co-O bond should be very strong, resulting in the symmetry effect not being significant on the RXES spectra. In the near future, we will perform a similar calculation with lower symmetry to further verify the symmetry effect.

#### 4. Conclusions

We have performed ex situ RXES at the Co L3 edge of LiCoO2 in combination with multiplet calculation and succeeded in separately clarifying the valence states of Co<sup>3+</sup> and Co<sup>4+</sup>. Both the Co<sup>3+</sup> and Co<sup>4+</sup> states show strong CT effects from the O 2p to Co 3d orbitals. The Co4+ state in the charged state has a negative CT energy, which means that the charge-transferred d<sup>6</sup>L configuration is dominant, resulting in an effective 3delectron number  $\langle n_{\rm d} \rangle$  of 5.9. Similar scenarios have been suggested in previous reports for other typical oxide-based cathode materials like LiMn<sub>2</sub>O<sub>4</sub>. The higher valence state in those oxides, such as the tetravalent state, commonly have strong CT effects. Thus, present and those results indicate that the O 2p orbital plays an essential role in the redox reaction through the strong p-d hybridization in oxide-based cathode materials. In addition, RXES in combination with difference analysis and multiplet calculation is a very powerful method to distinguish a specific oxidation state from mixed valence states. Extracting the higher valence state in the charged state of cathode materials is particularly important to understand the redox reaction. For example, control of the CT effect by some means, such as element substitution based on such electronic-structure analysis using RXES, will be effective to further improve the electrode performance.

#### Author contributions

D. A., E. H. and A. Y. conceived and directed the project. D. A. and E. H. synthesized and characterized LiCoO2. D. A., H. N., H. M. and E. H. constructed the operando cell. T. S., Y. N., R. W. and F. D. G. performed the multiplet calculations. D. A., T. S., H. K., K. Y., J. M., and Y. H. performed the soft X-ray absorption/emission spectroscopy. All of the authors wrote the manuscript.

### Data availability

The data that support the findings of this study are available from the corresponding author upon reasonable request.

#### Conflicts of interest

There are no conflicts to declare.

# Acknowledgements

This work was supported by the Ministry of Education, Culture, Sports, Science and Technology (MEXT), Japan; Grant-in-Aid for Specially Promoted Research (No. 15H05701). Part of this work was conducted on the basis of the International Joint Research Program for Innovative Energy Technology by the Ministry of Economy, Trade and Industry (METI), Japan. The XAS and RXES measurements at BL07LSU of SPring-8 were performed by the joint research in SRRO and ISSP, the University of Tokyo (Proposal No. 2017A7531, 2018B7588, 2019B7457, 2020A7477, 2021A7494, 2021A7495, 2021B7429, and 2022A7447).

#### References

- 1 K. Mizushima, P. C. Jones, P. J. Wiseman and J. B. Goodenough, Mat. Res. Bull., 1980, 15, 783-789.
- 2 K. Mizushima, P. C. Jones, P. J. Wiseman and J. B. Goodenough, Solid State Ionics, 1981, 3-4, 171-174.

- 3 J. N. Reimers and J. R. Dahn, J. Elecrochem. Soc., 1992, 139, 2091-2097.
- 4 J. van Elp, J. L. Wieland, H. Eskes, P. Kuiper, G. A. Sawatzky, F. M. F. de Groot and T. S. Turner, Phys. Rev. B: Condens. Matter Mater. Phys., 1991, 44, 6090-6103.
- 5 J. Sugiyama, H. Nozaki, J. H. Brewer, E. J. Ansaldo, G. D. Morris and C. Delmas, Phys. Rev. B: Condens. Matter Mater. Phys., 2005, 72, 144424.
- 6 M. Okubo, J. Kim, T. Kudo, H. S. Zhou and I. Honma, J. Phys. Chem. C, 2009, 113, 15337-15342.
- 7 B. Barbiellini, K. Suzuki, Y. Orikasa, S. Kaprzyk, M. Itou, K. Yamamoto, Y. J. Wang, H. Hafiz, R. Yamada, Y. Uchimoto, A. Bansil, Y. Sakurai and H. Sakurai, Appl. Phys. Lett., 2016, **109**, 073102.
- 8 Y. Uchimoto, H. Sawada and T. Yao, J. Synchrotron Radiat., 2001, 8, 872-873.
- 9 W.-S. Yoon, K.-B. Kim, M.-G. Kim, M.-K. Lee, H.-J. Shin, J.-M. Lee, J.-S. Lee and C.-H. Yo, J. Phys. Chem. B, 2002, 106, 2526-2532.
- 10 V. R. Galakhov, E. Z. Kurmaev, S. Uhlenbrock, M. Neumann, D. G. Kellerman and V. S. Gorshkov, Solid State Commun., 1996, 99, 221-224.
- 11 V. R. Galakhov, M. Neumann and D. G. Kellerman, Appl. Phys. A: Mater. Sci. Process., 2009, 94, 497-500.
- 12 V. R. Galakhov, N. A. Ovechkina, A. S. Shkvarin, S. N. Shamin, E. Z. Kurmaev, K. Kuepper, A. F. Takács, M. Raekers, S. Robin, M. Neumann, G.-N. Gavrila, A. S. Semenova, D. G. Kellerman, T. Käämbre and J. Nordgren, Phys. Rev. B: Condens. Matter Mater. Phys., 2006, 74, 045120.
- 13 D. Ensling, A. Thissen, S. Laubach, P. C. Schmidt and W. Jaegermann, Phys. Rev. B: Condens. Matter Mater. Phys., 2011, 82, 195431.
- 14 T. Mizokawa, Y. Wakisaka, T. Sudayama, C. Iwai, K. Miyoshi, J. Takeuchi, H. Wadati, D. G. Hawthorn, T. Z. Regier and G. A. Sawatzky, Phys. Rev. Lett., 2013, 111, 056404.
- 15 C. J. Partridge, C. T. Love, K. E. Swider-Lyons, M. E. Twigg and D. E. Ramaker, J. Solid State Chem., 2013, 203, 134-144.
- 16 J. M. Rosolen, P. Ballirano, M. Berrettoni, F. Decker and M. Gregorkiewitz, Ionics, 1997, 3, 345-354.
- 17 A. Kotani and S. Shin, Rev. Mod. Phys., 2001, 73, 203-246.
- 18 L. J. Ament, M. van Veenendaal, T. P. Devereaux, J. P. Hill and J. van den Brink, Rev. Mod. Phys., 2011, 83, 705-767.
- 19 F. de Groot and A. Kotani, Core level spectroscopy of solids, CRC Press, Boca Raton, Florida, 2008.
- 20 D. Asakura, Y. Nanba, Y. Makinose, H. Matsuda and E. Hosono, ChemPhysChem, 2018, 19, 988-992.
- 21 D. Asakura, Y. Nanba, Y. Makinose, H. Matsuda and E. Hosono, Phys. Chem. Chem. Phys., 2017, 19, 16507-16511.
- 22 D. Asakura, Y. Nanba, E. Hosono, M. Okubo, H. Niwa, H. Kiuchi, J. Miyawaki and Y. Harada, Phys. Chem. Chem. Phys., 2019, 21, 18363-18369.
- 23 T. Sudayama, K. Uehara, T. Mukai, D. Asakura, X.-M. Shi, A. Tsuchimoto, B. Mortemard de Boisse, T. Shimada, E. Watanabe, Y. Harada, M. Nakayama, M. Okubo and A. Yamada, Energy Environ. Sci., 2020, 13, 1492-1500.

- 24 W. E. Gent, K. Lim, Y. F. Liang, Q. H. Li, T. Barnes, S. J. Ahn, K. H. Stone, M. McIntire, J. Y. Hong, J. H. Song, Y. Y. Li, A. Mehta, S. Ermon, T. Tyliszczak, D. Kilcoyne, D. Vine, J. H. Park, S. K. Doo, M. F. Toney, W. L. Yang, D. Prendergast and W. C. Chueh, Nat. Commun., 2017, 8, 2091.
- 25 K. Luo, M. R. Roberts, R. Hao, N. Guerrini, D. M. Pickup, Y.-S. Liu, K. Edström, J.-H. Guo, A. V. Chadwick, L. C. Duda and P. G. Bruce, Nat. Chem., 2016, 8, 684-691.
- 26 R. A. House, U. Maitra, M. A. Pérez-Osorio, J. G. Lozano, L. Jin, J. W. Somerville, L. C. Duda, A. Nag, A. Walters, K.-J. Zhou, M. R. Roberts and P. G. Bruce, Nature, 2020, 577, 502-508.
- 27 R. A. House, G. J. Rees, M. A. Pérez-Osorio, J. G. Lozano, J.-J. Marie, E. Boivin, A. W. Robertson, A. Nag, M. Garcia-Fernandez, K.-J. Zhou and P. G. Bruce, Nat. Energy, 2020, 5, 777-785.
- 28 R. A. House, J.-J. Marie, M. A. Pérez-Osorio, G. J. Rees, E. Boivin and P. G. Bruce, Nat. Energy, 2021, 6, 781-789.
- 29 L. Massel, K. Hikima, H. Rensmo, K. Suzuki, M. Hirayama, C. Xu, R. Younesi, Y.-S. Liu, J.-H. Guo, R. Kanno, M. Hahlin and L.-C. Duda, J. Phys. Chem. C, 2019, 123, 28519-28526.
- 30 J.-N. Zhang, Q. Li, C. Ouyang, X. Yu, M. Ge, X. Huang, E. Hu, C. Ma, S. Li, R. Xiao, W. Yang, Y. Chu, Y. Liu, H. Yu, X.-Q. Yang, X. Huang, L. Chen and H. Li, *Nat. Energy*, 2019, 4, 594–603.
- 31 E. Hu, Q. Li, X. Wang, F. Meng, J. Liu, J.-N. Zhang, K. Page, W. Xu, L. Gu, R. Xiao, X. Huang, L. Chen, W. Yang, X. Yu and X.-Q. Yang, Joule, 2021, 5, 720-736.
- 32 S. Yamamoto, Y. Senba, T. Tanaka, H. Ohashi, T. Hirono, H. Kimura, M. Fujisawa, J. Miyawaki, A. Harasawa, T. Seike, S. Takahashi, N. Nariyama, T. Matsushita, M. Takeuchi, T. Ohata, Y. Furukawa, K. Takeshita, S. Goto, Y. Harada, S. Shin, H. Kitamura, A. Kakizaki, M. Oshima and I. Mastuda, J. Synchrotron Radiat., 2014, 21, 352-365.
- 33 Y. Harada, M. Kobayashi, H. Niwa, Y. Senba, H. Ohashi, T. Tokushima, Y. Horikawa, S. Shin and M. Oshima, Rev. Sci. Instrum., 2012, 83, 013116.
- 34 M. W. Haverkort, M. Zwierzcki and O. K. Andersen, Phys. Rev. B: Condens. Matter Mater. Phys., 2012, 85, 165113.
- 35 R.-P. Wang, B. Liu, R. J. Green, M. U. Delgado-Jaime, M. Ghiasi, T. Schmitt, M. M. van Schooneveld and F. M. F. de Groot, J. Phys. Chem. C, 2017, 121, 24919-24928.
- 36 H.-J. Kim, Y. U. Jeong, J.-H. Lee and J.-J. Kim, J. Power Sources, 2006, 159, 233-236.
- 37 D. Qian, Y. Hinuma, H. Chen, L.-S. Du, K. J. Carroll, G. Ceder, C. P. Grey and Y. S. Meng, J. Am. Chem. Soc., 2012, 134, 6096-6099.
- 38 L. A. Montoro, M. Abbate and J. M. Rosolen, Electrochem. Solid State Lett., 2000, 3, 410-412.
- 39 R. H. Potze, G. A. Sawatzky and M. Abbate, Phys. Rev. B: Condens. Matter Mater. Phys., 1995, 51, 11501-11506.
- 40 Y. Yamauchi and H. Nakai, J. Electrochem. Soc., 2013, 160, A1364-A1368.
- 41 D. Asakura, E. Hosono, H. Niwa, H. Kiuchi, J. Miyawaki, Y. Nanba, M. Okubo, H. Matsuda, H. S. Zhou, M. Oshima and Y. Harada, Electrochem. Commun., 2015, 50, 93-96.
- 42 Y. Nanba, T. Iwao, B. Mortemard de Boisse, W. Zao, E. Hosono, D. Asakura, H. Niwa, H. Kiuchi, J. Miyawaki,

- Y. Harada, M. Okubo and A. Yamada, Chem. Mater., 2016, 28, 1058-1065.
- 43 M. Imada, A. Fujimori and Y. Tokura, Rev. Mod. Phys., 1998, 70, 1039-1263.
- 44 A. Fujimori and F. Minami, Phys. Rev. B: Condens. Matter Mater. Phys., 1984, 30, 957-971.
- 45 K. Kawai, D. Asakura, S.-I. Nishimura and A. Yamada, Chem. Commun., 2019, 55, 13717-13720.