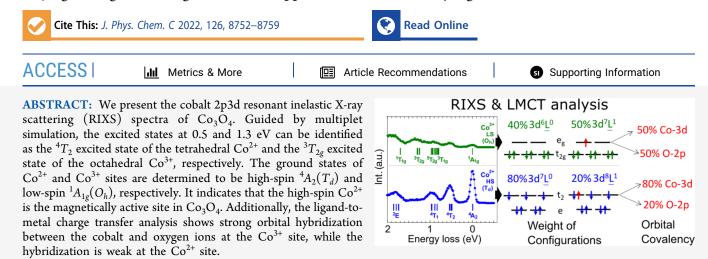


pubs.acs.org/JPCC

Analyzing the Local Electronic Structure of Co₃O₄ Using 2p3d Resonant Inelastic X-ray Scattering

Ru-Pan Wang, Meng-Jie Huang, Atsushi Hariki, Jun Okamoto, Hsiao-Yu Huang, Amol Singh, Di-Jing Huang, Peter Nagel, Stefan Schuppler, Ties Haarman, Boyang Liu,* and Frank M. F. de Groot*



1. INTRODUCTION

Electronic properties of 3d transition-metal (TM) oxides are governed by the interactions between the TM ions and the neighboring contributions. Those interactions are determined by Hund's coupling, crystal-field splitting, spin—orbit coupling, and the interatomic hybridization (covalent bonding). The determination of the local electronic structure is a relatively straightforward task for localized materials, but it faces an experimental challenge for a mixed-valence TM compound, especially when the constituting TM ions have a rich overlapping multiplet structure, which is the case for Co_3O_4 .

 Co_3O_4 crystallizes in a normal spinel structure (AB₂O₄) below 850 K.^{1,2} The Co ions in the A sites with tetrahedral (T_d) symmetry are divalent (Co²⁺), while those in the B sites with octahedral (O_h) symmetry are trivalent (Co^{3+}) with a small trigonal distortion. An experimental determination of the electronic structure of two inequivalent Co sites is of fundamental importance to elucidate the magnetic coupling responsible for the antiferromagnetic order (below ~40 K).^{3,} The Co-site dependence of covalency is crucial for large exchange anisotropies achieved by a substitution of Ni or Fe for Co ions and its application to magneto-optical information storage.^{5–7} The covalency also influences the charge capacity. Co_3O_4 shows a high charge capacity (700 mAh/g) and a good cycle performance of nanosize negative electrodes, which makes Co₃O₄ an interesting compound for the lithium battery applications.^{8,9} Moreover, the role of Co²⁺ ions in the oxygen evolution reaction step of water oxidation is debated.¹⁰⁻¹⁷ A small band gap was found in Co_3O_4 that is important for the application in a photovoltaic cell,^{18–20} where the electronic structures of individual sites potentially plays an important role.

The electronic structure of Co_3O_4 has been characterized by the ultraviolet/visible $(\text{UV/vis})^{21-25}$ and near-infrared (NIR) absorption.^{4,26} These absorption spectra show excitations at 0.82, 0.93, 1.64, and 2.81 eV,²² while diverse interpretations for these excitations are proposed,^{4,10,21-25} partly related to the fact that the dd excitations are usually dipole forbidden.²⁷ Hibberd et al. revealed distinct absorption peaks for the Co valence (2+ or 3+) and local environment (O_h or T_d) of Co_3O_4 using Co L_{2,3}-edge (2p) X-ray absorption spectroscopy (XAS).²⁸ However, due to possible overlap of the multiple Co-site signals and the lifetime broadening (~200 meV), the details of the local electronic structure are limited.

In this work, we study the Co-site resolved low-energy local excitations in Co_3O_4 using Co 2p3d resonant inelastic X-ray scattering (RIXS). By setting the resonant photon energy to the distinct Co L₃-XAS features of the Co^{2+} and Co^{3+} sites,²⁸ the site-selective local excitations are measured. The site-selectivity of RIXS has been discussed in literature.^{29–31} The (photon-in-photon-out) transitions in RIXS brings a broad sensitivity to dd excitations of the system that provides a better determination of the local electronic structure including the orbital hybridization effects.^{32–36} This has also been

 Received:
 March 3, 2022

 Revised:
 April 27, 2022

 Published:
 May 11, 2022





demonstrated using 1s3d RIXS on Co_3O_4 ,^{37,38} but the intensity was limited by the absorption cross-section of the quadrupole 1s3d transition. We show that the Co 2p3d RIXS is a good chemical site-selective approach to study accurate details of the ground state and the core excited states.

2. METHODOLOGY

2.1. Experimental Details and Sample Preparation. The experiments were performed on 99.9985% Co₃O₄ powder produced by Alfa Aesar. A 10 mm × 0.5 mm (diameter × height) cylindrical pellet Co₃O₄ was prepared for the measurements. Co 2p XAS spectra were acquired at the soft X-ray WERA beamline of the Karlsruhe Research Accelerator (KARA) synchrotron in Germany. The instrumental resolution was calibrated to be ~280 meV full width at half-maximum (fwhm) at the Co $L_{2,3}$ edge (~780 eV). Both the total electron yield (TEY) and the fluorescence yield (FY) methods were employed. The Co 2p3d RIXS measurements with linearly vertical (V) and horizontal (H) polarized incident X-rays were performed at the Taiwan Light Source (TLS) beamline 05A of National Synchrotron Radiation Research Center (NSRRC) in Taiwan.³⁹ The experimental energy resolution of the incident photon was ~700 meV and the combined resolution of RIXS was ~90 meV. A grazing incident geometry (~ 20°) with the spectrometer at 90° was used. Given the energy-compensation principle, the RIXS resolution is much better than the incident photon resolution.^{39,40} For a perfect monochromator-spectrometer pair, the photons will be focused at the mirror position of the photon source. As soon as the photon lost its energy in the path, the focus will not be kept at the same position. Thus, using a position sensitive detector combined with the active monochromator-spectrometer system, the RIXS resolution can be decoupled from the incident photon resolution. To compare the XAS spectra measured at different facilities, the background signals were subtracted from the spectra, as described in the Supporting Information (SI). The subtracted spectra were normalized to the maximum of the Co L₃ edge. We calibrated the photon energy of the RIXS to the absorption spectra acquired at the WERA beamline. The RIXS spectra measured with horizontal (H) polarization were normalized to the exposure time. Then the spectra with vertical (V) polarization were normalized to the H-polarization spectra according to the profile at high energies (above 2.5 eV). The measurements at the KARA-WERA and TLS-05A beamlines were carried out at 50 and 150 K, respectively. We note that the simulated intensity is given as absolutely crosssection, while the experimental intensity is given in normalized units.

2.2. Theory. The results were analyzed using a cluster model that includes the Coulomb multiplet interaction, the crystal-field splitting, the spin—orbit coupling, and the charge transfer between Co 3d and O 2p orbitals. The calculation was carried out with the program Quanty, which implements a configuration-interaction scheme.^{41,42} To study the mixed-valence spectra of Co₃O₄, the Co²⁺(T_d) site and the Co³⁺(O_h) site were calculated separately. We took the initial parameters from literature for a high-spin Co²⁺ ground state (⁴ B_1 in D_{2d} symmetry) and a low-spin Co³⁺ ground state (¹ A_{1g} in O_h symmetry).⁴³⁻⁴⁵ The values were refined by a detailed comparison with the high-resolution RIXS data and its polarization dependence (section 3.2). To evaluate the used values, we compared the ones estimated by an ab initio calculation for Co₃O₄. The ab initio calculation was based on a

density-functional calculation with the local-density approximation (LDA). The parameters (crystal field and hopping parameters) were extracted from the tight-binding model which spans the Co 3d and O 2p bands by a Wannier projection.

3. RESULTS

3.1. 2p3d RIXS Experimental Results. The experimental Co 2p XAS and 2p3d RIXS results are presented in Figure 1.

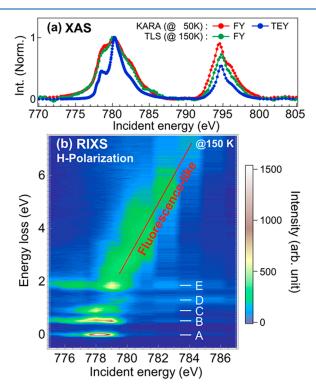


Figure 1. Experimental results. (a) The H-polarization 2p XAS spectra of Co_3O_4 . (b) The H-polarization 2p3d RIXS energy map of Co_3O_4 .

The TEY spectrum shows sharp features at 778.5 and 780.2 eV, which are the characteristic signals of the Co^{2+} and the Co^{3+} sites in Co_3O_4 , respectively.²⁸ The FY spectrum, on the other hand, is rather broad, and the features are obscure. This is due to strong saturation and self-absorption effects for a bulk sample. In the 2p3d RIXS result (Figure 1b), a fluorescence-like signal increases with incident photon energies as guided by a red line. Sharp features at ~0.0, 0.5, 0.9, 1.2, and 1.9 eV are observed and labeled from A to E, respectively. The features A–C are resonantly enhanced at 778 eV, while feature D is enhanced at 780 eV. Feature E resonates at 778.7 eV. These features are attributed to local excitations of the Co^{2+} and Co^{3+} sites, as we will discuss later.

To gain more insight, Figure 2a compares the H- and Vpolarization energy loss spectra at three different incident energies. At 780.0 eV, feature C (at ~0.9 eV) is enhanced by the V-polarization, while feature E (at ~1.9 eV) shows opposite behavior. This polarization dependence resembles the ${}^{3}T_{1g}$ and ${}^{1}T_{1g}$ excited states in LaCoO₃ at 20 K, where the Co³⁺ ions have the ${}^{1}A_{1g}$ ground state, see Figure 2a,b. The energy differences indicate that the crystal field splitting varies between the two. Feature B (at ~0.5 eV) is identified as the ${}^{4}T_{2}$ excited state of the Co²⁺(T_{d}) site.⁵ As a reference, divalent

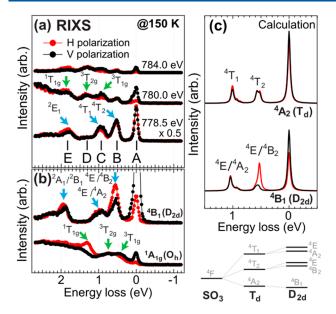


Figure 2. Comparison of H- and V-polarization RIXS spectra of (a) the Co_3O_4 and (b) the ${}^{4}B_1(D_{2d})$ and ${}^{1}A_{1g}(O_h)$ ground states from refs 44 and 45. The blue (green) arrows indicate the characteristic features of the $Co^{2+}(Co^{3+})$ site. (c) The calculated 2p3d RIXS polarization comparison of distorted and nondistorted tetrahedral Co^{2+} using parameters in ref 44 without considering the ligand-to-model charge transfer.

 ${}^{4}B_{1}(D_{2d} \text{ symmetry}) \text{ Co}^{2+}$ RIXS at the L₃ edge of K₅H-[CoW₁₂O₄₀]·*x*H₂O is shown in Figure 2b,⁴⁴ where the local structure is a distorted tetrahedral Co²⁺. The distortion changes the ground state symmetry from the ${}^{4}A_{2}(T_{d})$ to the ${}^{4}B_{1}(D_{2d})$ that gives rise to a strong polarization dependence at the 0.5 eV peak in the RIXS spectra. However, the feature B shows no dichroism in Co₃O₄, which suggests that the distortion is negligibly small on the Co²⁺ site and supports the ${}^{4}A_{2}(T_{d})$ symmetry of the ground state in the calculation.

3.2. Multiplet Simulation Results. Guided by the experimental results, the low-energy local excitations characterize the two Co sites in Co₃O₄ and the crystal-field parameters can be determined. In the cluster-model analysis, these parameters identify the symmetry of the local site and the crystal-field energy (10Dq) is the key factor to identify the energy splitting of the Co 3d orbitals $(t_{2g}(t_2) \text{ and } e_g(e) \text{ orbitals}$ in $O_h(T_d)$ symmetry). For a system involving charge transfer effects, the crystal-field energy can be decomposed into two components:³⁶ the ionic crystal field energy $(10Dq_{ionic})$ and the effective crystal field energy induced by the charge transfer effect $(10Dq_{CT})$. The combination of the two $(10Dq_{tot})$ gives the final splitting between the e/e_g and t/t_{2g} orbitals. Figure 3b,c shows the excited states energies as a function of the $10Dq_{\text{ionic}}$ for the Co³⁺ and Co²⁺ site, respectively. The 2p3d RIXS probed at 780.0 eV shows three features at 0.9, 1.3, and 1.9 eV, indicated with the green arrows in Figure 3a, which are attributed to the signals at the Co³⁺ site. The feature at 0.5 eV shows no polarization dependence, which indicates that the feature is likely the tail contribution of the ${}^{4}T_{2}$ feature of the Co^{2+} site (cf Figure 2). As seen in Figure 3b, the matching of the three features gives $10Dq_{ionic} \sim 1.15$ eV at the Co³⁺ site. We note that this value is smaller than the $10Dq_{tot}$ value due to the missing of $10Dq_{CT}$. The $10Dq_{tot}$ values will be given in Table 1. Further discussion about the $10Dq_{tot}$ can be found in the SI. The $10Dq_{ionic}$ value is much larger than the spin-state



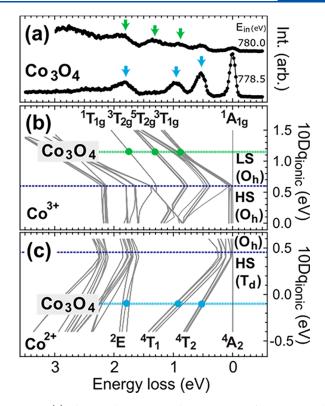


Figure 3. (a) The H-polarization 2p3d RIXS spectra of Co_3O_4 excited at 778.5 and 780.0 eV. The calculated energy diagrams of (b) the Co^{3+} ion and (c) the Co^{2+} ion as a function of ionic crystal field energy $(10Dq_{\text{ionic}})$, including the charge transfer.

Table 1. Model Parameters Used in the Simulation (in eV), Which Are the Crystal Field Energy, Hopping Integrals, Charge Transfer Energy, U_{dd} and $U_{pd}^{\ a}$

	$10Dq_{ionic}$	$10Dq_{tot}$	Δ	$V_{\rm e(e_g)}$	$V_{\mathrm{t_2(t_{2g})}}$	$U_{\rm dd}$	$U_{ m pd}$
Co_i^{2+}	-0.10	-0.55	4.5	1.0	2.0	4.5	
Co_m^{2+}	-0.02	-0.47	4.5	1.0	2.0	4.5	6.0
Co_i^{3+}	1.15	1.90	1.5	3.12	1.8	6.5	
Co_m^{3+}	0.84	1.59	1.5	3.12	1.8	6.5	7.5

^{*a*}The i and m stand for the configurations of initial ground state and intermediate state, respectively.

transition point $(10Dq_{\text{ionic}} \sim 0.6 \text{ eV})$, suggesting that the ground state on the Co³⁺ site in Co₃O₄ is a robust low-spin singlet ${}^{1}A_{1g}$. For the Co²⁺ site, the $10Dq_{\text{ionic}} \sim -0.1 \text{ eV}$ reproduces the energy position of the characteristic features at 0.5, 0.9, and 1.9 eV in the RIXS data probed at 778.5 eV, which are indicated in Figure 3a,c. The negative 10Dq value indicates an inversion of the e and t₂ manifolds in the T_d symmetry.

Table 1 summarizes the key parameters, and more values can be found in SI. The charge transfer energy Δ is an energy related to electron transfer from a ligand to the Co site and $V_{\rm e_{(g)}}/V_{\rm t_{2(g)}}$ are the values for electron hopping. The $U_{\rm dd}$ and $U_{\rm pd}$ values parametrize the Coulomb interaction, which are set to reference values.^{36,43} Since the ligand-to-metal charge transfer also contributes to the energy splitting of Co 3d states,³⁶ we provide the $10Dq_{\rm tot}$ value that is calculated by the resultant energy splitting by including the charge transfer. To simulate a contraction of Co 3d wave functions by the presence of the core hole,⁴⁶ the $10Dq_{\rm tot}$ value in the intermediate state was reduced from that in the ground state by ~15%, see also SI.

To evaluate the parameters in Table 1, the values estimated by the LDA calculation are provided. For the $Co^{2+}(T_d)$ site, the estimated values are -0.10 eV for $10Dq_{ionic}$ and 1.29 (1.82) eV for V_e (V_{t_2}). For the Co³⁺(O_h) site, the estimated values are 0.7 eV for 10D q_{ionic} and 3.03 (1.74) eV for $V_{e_{q}}$ ($V_{t_{2q}}$, where we omitted the energy splitting within the $t_{\rm 2g}$ manifolds due to a small trigonal distortion and estimated the $V_{t_{2r}}$ value by averaging the hopping integrals over the two. The $D\sigma$ value, which measures the trigonal distortion, is estimated as 0.05 eV. This value is much smaller than the required value (>0.5 eV)for changing the ground state symmetry (singlet ${}^{1}A_{1\sigma}$) of the Co^{3+} site in Co_3O_4 , but gives a minor correction to the multiplet energies. Because the trigonal distortion gives no visible effect on the spectra, we neglect it in our simulation and the site is referred to as O_h for simplicity. We conclude that the optimized values agree well with the ab initio estimates except for a small discrepancy in the $10Dq_{ionic}$ value at the Co³⁺ site, which is possibly due to an underestimate of the covalency in the LDA scheme. 42,47

Figure 4a,b shows the RIXS calculation of the Co^{2+} and Co^{3+} sites with the values listed in Table 1. Since the low-spin $({}^{1}A_{1\rho})$

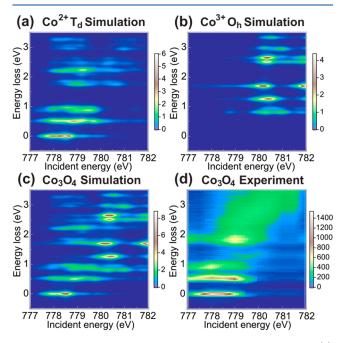


Figure 4. Simulated H-polarization 2p3d RIXS energy maps. (a) $\operatorname{Co}^{2+4}A_2(T_d)$ ground state. (b) $\operatorname{Co}^{3+1}A_{1g}(O_h)$ ground state. (c) The simulation of Co_3O_4 . (d) Experiment of Co_3O_4 .

ground state on the Co³⁺ site is angular isotropic, its intensity is strongly suppressed when the scattering angle is about 90° with the H-polarization condition. Thus, the intensity at zero energy loss (elastic line) is mainly due to the Co²⁺ site with the ⁴A₂ ground state. Figure 4c shows the sum of the calculated intensities which was corrected by a combination ratio of stoichiometry (1:2) and the number of holes (3:4), where the ratio was also applied to weight the area normalized XAS spectra. The incident energies of Co²⁺ and Co³⁺ spectra were adjusted by a 1.8 eV difference in between the main L₃ feature to reproduce the experimental Co L₃ XAS (cf Figure 5a). This value agrees with the finding in literature (1.7 eV).²⁸ The energy shift is not only depending on the oxidation state but also on the spin-state, symmetry (O_h, respectively, T_d) and

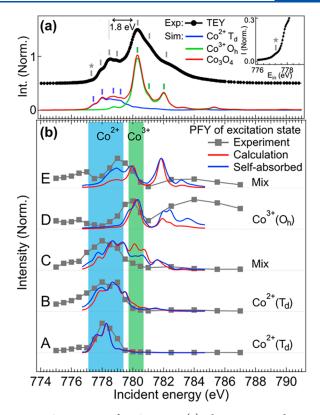


Figure 5. Comparison of XAS spectra. (a) The comparison between the experimental TEY spectrum and the simulations. The inset panel zooms in the pre-edge region and indicates a small feature before the edge. This small feature matches to the first feature indicated in the simulation. (b) The comparison of the PFY spectra correspond to the excited states A–E.

hybridization (charge transfer effect). Some intensity discrepancies are observed, that is, the calculated RIXS at 780.5 eV shows much stronger 1.3, 1.7, and 2.5 eV excited features. These discrepancies can be explained by saturation and selfabsorption effects (see section 4.1) and further related to the differential broadening at different site (see section 4.2).

4. DISCUSSION

4.1. Site Selectivity of Partial Fluorescence Yield Spectra. Partial fluorescence yield (PFY) spectra were acquired for energy transfer of the dd excitations (A-E) from the RIXS map to discuss the local excitations in detail. The experimental data are obtained by fitting the RIXS intensities of these dd excitations (cf Figure 4d), and the theoretical spectra are obtained by summing simulated intensities in an energy window larger than the width of these excitations in Figure 4c. Since the features are well separated in the simulation, this summation represents the peak intensity properly.

Before the discussion on the PFY spectra, the TEY result is compared ahead with the simulated spectra. In Figure 5a, the sum of the simulated XAS of Co^{2+} (blue) and Co^{3+} (green) sites well matches with the experimental TEY data (gray bars and red line). The contributions of two sites are separated in the incident energies that enables to observe site-resolved local excitations by RIXS, as we show in Figure 4. The site selectivity can be better understood from the PFY spectra. In the PFY spectra (Figure 5b), the features A–B and D are unambiguously attributed to the Co^{2+} site and the Co^{3+} site excitations, respectively, while the features C and E show an overlap with both Co^{2+} and Co^{3+} regions. We stress that the overlap is essential since, according to our simulation in Figure 4a,b, both sites have excitations at around 0.9 and 1.9 eV. A systematic discrepancy is observed that the intensity at about 780 eV is always overestimated in the simulation, particularly, of the PFY spectra of the features C and E. To analyze the two features, the saturation and self-absorption effects need to be applied and the correction can be written as^{48,49}

$$I_{\rm ems} = I_0 \frac{S_X(\omega_{\rm in}\omega_{\rm out})\mu_X(\omega_{\rm in})}{\mu_{\rm tot}(\omega_{\rm in}) + \mu_{\rm tot}(\omega_{\rm out})\frac{\sin\theta}{\sin(\alpha-\theta)}}$$

Here, X, α , and θ refer to the emission edge of element, the scattering angle, and the sample rotation angle, respectively. In this work, the experimental geometry was set to $\alpha = 90^{\circ}$ and θ = 20°. The $\mu_{tot}(\omega_{in})$ and $\mu_{tot}(\omega_{out})$ are the absorption factor of the photon-in and photon-out channel in the RIXS process. We multiplied a self-absorption coefficient $\left(\mu_{\text{tot}}(\omega_{\text{in}}) + \mu_{\text{tot}}(\omega_{\text{out}})\frac{\sin\theta}{\sin(\alpha-\theta)}\right)^{-1}$ to the simulated RIXS result (multiplication of $S_X(\omega_{in}, \omega_{out})$ and $\mu_X(\omega_{in})$). The formula implies that the saturation effect is stronger for a large absorption factor. Thus, the features at ~780 eV in PFY spectra are to be suppressed. Consequently, the PFY weights on the Co^{3+} region is largely suppressed for the feature C (cf. Figure 5b). It also enhances the intensity at ~782 eV of PFY spectra for the feature D and shows better agreement with the experiment. However, there is still an intensity discrepancy of the feature at 782 eV for excitation E. This overestimation of 782 eV feature is due to the model of oxygen 2p orbitals, where its bandwidth is neglected. In our simulations, the energy broadening was assumed to be the same for both sites, which yields rather sharp features in the Co³⁺ site contribution compared to the experiment. A larger energy broadening for the Co3+ site is caused by the strong ligand-metal hybridization on the Co³⁺ site. To reproduce the broad XAS/PFY structure above the L₃ edge, the band formation of ligand 2p states must be taken into accounted, which is beyond the description of the cluster model used in this study.

The present analysis shows that the excited states at 0.5 and 1.3 eV are the unique features to identify the ${}^{4}T_{2}(T_{d})$ excited state on the Co²⁺ site and ${}^{3}T_{1g}(O_{h})$ excited state on the Co³⁺ site. In comparison with the infrared⁴ and 1s3d RIXS, 37,38 those individual site can be resolved much better. Our result confirms that the Co₃O₄ is mainly composed by the magnetically active high-spin Co²⁺(T_{d}) and the diamagnetic low-spin Co³⁺(O_{h}).

4.2. Ligand–Metal Hybridization Influence of Different Co Sites in Co₃O₄. We discuss here the question whether the ligand–metal hybridization influences the local electronic structure based on the ligand-to-metal charge transfer model. Hibberd et al. have shown in their ionic-model analysis that the Slater integrals of the ionic Coulomb multiplet need to be reduced substantially to reproduce the Co L₃ XAS spectra, which implies that the ligand–metal hybridization is strong in Co₃O₄.^{18,28,50} The sensitivity of 2p3d RIXS to the dd excitations allows us to address the question, and furthermore study the site-dependence of the covalency in Co₃O₄. Figure 6 compares the energy diagrams obtained with three different models: (i) the ionic model with the bare (ionic) values of the Slater integrals, (ii) the one with reduced Slater integrals, and (iii) the cluster model including the ligand-to-metal charge

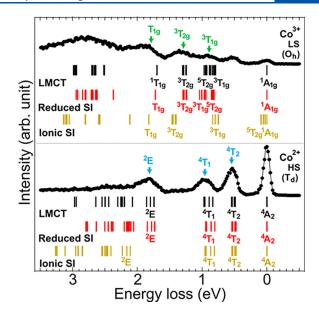


Figure 6. Energy diagrams calculated using the values of the ionic Slater integrals (ionic SI), the reduced Slater integrals (reduced SI), and the ligand-to-metal charge transfer (LMCT) effect.

transfer channel explicitly. Apparently, the ionic model (i) overestimates energies of observed excitations. For the Co^{2+} site, the ionic model (ii) with reduced Slater integrals (80% from the ionic values) yields good agreement with the experimental data. Note that both model (i) and (ii) show nice agreement on the ${}^{4}T_{2}$ and ${}^{4}T_{1}$ excited states, which indicates that those features are less sensitive to the hybridization change. On the other hand, for the $Co^{3+}(O_{h})$ site, the Slater integrals are reduced to 55% (80%) from the ionic F^{2}_{dd} (F^{4}_{dd}) value to fit the experimental data. These unconventional reduction rates for the $Co^{3+}(O_{h})$ site imply that the differential screening effect via a ligand-to-metal charge transfer channel is not negligible. The cluster model including the ligand-to-metal charge transfer channel shows good agreement to the experimental results.

To obtain further information about the ligand-metal hybridization, Table 2 shows the configuration weights in the

Table 2. Weight of Configurations and Cation Orbital Covalency in the Ground State (unit in %)^{*a*}

	we	ight of configu	orbital covalency		
	$ 3d^n\rangle$	$ 3d^{n+1}\underline{L}^1\rangle$	$ 3d^{n+2}L^2\rangle$	e(e _g)	$t_2(t_{2g})$
$Co^{2+}(3d^7)$	79	20	1	100	80
$Co^{3+}(3d^6)$	40	50	10	50	100

^{*a*}Although the number of ligand holes is considered up to two in the spectral simulations, the covalency is estimated only using the configurations up to one ligand hole.

ground state of the cluster model using the optimized parameters in Table 1. The ligand-to-metal charge transfer channel mixes the ionic configuration $(3d^n)$ with ones with ligand holes $(3d^{n+1}\underline{L}^1 \text{ and } 3d^{n+2}\underline{L}^2)$. In the Co³⁺ site, the ligand-hole configurations $(3d^7\underline{L}^1 \text{ and } 3d^8\underline{L}^2)$ show large weights, which indicates that the Co³⁺ site is strongly hybridized with the ligand states. In contrast, the ligand-hole configuration $(3d^8\underline{L}^1)$ contributes only ~20% to the ground state in the Co²⁺ site. This observation suggests that the Co³⁺ site is highly covalent, while the Co²⁺ site is rather ionic in Co₃O₄. The

difference also affects the RIXS profile: the ionic Co^{2+} site shows sharp local dd excitations; the covalent Co^{3+} site exhibits a broad intense fluorescence-like feature (cf. Figure 1).

The orbital covalency of the Co^{2+} (Co^{3+}) cation was analyzed using the approach described in the SI³⁶ and is summarized in Table 2. The orbital covalencies of e and t_2 orbitals on the tetrahedral Co^{2+} cation in Co_3O_4 are 100% and 80%, respectively. The e orbital is fully occupied and cannot participate in the ligand-metal hybridization, thus 100% orbital covalency is found. A high value of the cation orbital covalency for t₂ orbital indicates that it less contributes to the ligand-hole configuration $d^{8}L$, which is consistent with the ionic character of the Co^{2+} site. For the Co^{3+} site, the orbital covalencies of e_g and t_{2g} orbitals are 50% and 100%. This indicates that the ligand-metal hybridization mainly influences to the e_g orbital of the Co³⁺ ions. Although the Co³⁺ cation is the singlet ground state ($t_{2g}^{6}e_g^{0}$ state in the ionic picture), the e, orbital forms a strong bonding with neighboring oxygen 2p orbitals. This bonding is represented by the $d^{7}L$ configuration in an e_g symmetry, yielding the reduced value (50%) of the orbital covalency.

5. CONCLUSIONS

We present the Co 2p XAS and 2p3d RIXS experimental results in comparison with cluster model simulations on Co₃O₄. The 2p3d RIXS provides good chemical site selectivity to the local electronic structure, from which we can identify orbital covalencies of different ions in the compound. The polarization-dependent analysis indicates the symmetry character of the dd excitations, which provides a solid guide to analyze the local electronic structure. By selecting characteristic excitations, the PFY spectra are able to give additional site dependent information. The result shows the ${}^{4}T_{2}$ excited state of the tetrahedral Co²⁺ site at 0.5 eV, which is beyond the discriminative power of optical absorption. In addition, the ${}^{1}A_{1g}$ to ${}^{3}T_{2g}$ excitation of the octahedral Co³⁺ site at 1.3 eV can be uniquely identified. The ground state electronic structure of the Co^{2+} ions and the Co^{3+} ions are respectively high-spin ${}^{4}A_{2}(T_{d})$ and low-spin ${}^{1}A_{1g}(O_{h})$, where the high-spin Co²⁺ must be the magnetically active site. Our result also shows strong ligand-metal hybridization on the Co³⁺ site, which indicates that the Co^{3+} site in Co_3O_4 is rather covalent. In contrast, the Co^{2+} site shows weak hybridization implying that Co^{2+} is more ionic. This chemical site selectivity will help the further understanding on the site-dependent catalytic activity and magnetic activity of the spinel oxides.

ASSOCIATED CONTENT

Supporting Information

The Supporting Information is available free of charge at https://pubs.acs.org/doi/10.1021/acs.jpcc.2c01521.

Details about background subtraction of the XAS spectra (section A), an estimation of the theoretical absorption background (section B), all used parameters (section C), the comparison of the simulated XAS spectra (section D), and an explanation of the orbital covalency (section E) (PDF)

AUTHOR INFORMATION

Corresponding Authors

Boyang Liu – Debye Institute for Nanomaterials Science, Utrecht University, 3584 CG Utrecht, The Netherlands; Phone: (+31) 302537400; Email: boyang_liu@ vip.sina.com

Frank M. F. de Groot – Debye Institute for Nanomaterials Science, Utrecht University, 3584 CG Utrecht, The Netherlands; © orcid.org/0000-0002-1340-2186; Phone: (+31) 302537400; Email: f.m.f.degroot@uu.nl

Authors

- Ru-Pan Wang Debye Institute for Nanomaterials Science, Utrecht University, 3584 CG Utrecht, The Netherlands; Department of Physics, University of Hamburg, 22761 Hamburg, Germany; © orcid.org/0000-0003-4495-9881
- Meng-Jie Huang Karlsruhe Institute of Technology, D-76021 Karlsruhe, Germany; Deutsches Elektronen-Synchrotron DESY, 22607 Hamburg, Germany
- Atsushi Hariki Department of Physics and Electronics, Graduate School of Engineering, Osaka Prefecture University, Sakai, Osaka 599-8531, Japan
- Jun Okamoto National Synchrotron Radiation Research Center, Hsinchu 30076, Taiwan
- Hsiao-Yu Huang National Synchrotron Radiation Research Center, Hsinchu 30076, Taiwan
- Amol Singh National Synchrotron Radiation Research Center, Hsinchu 30076, Taiwan; Occid.org/0000-0003-0966-5132
- Di-Jing Huang National Synchrotron Radiation Research Center, Hsinchu 30076, Taiwan
- **Peter Nagel** Karlsruhe Institute of Technology, D-76021 Karlsruhe, Germany
- Stefan Schuppler Karlsruhe Institute of Technology, D-76021 Karlsruhe, Germany
- **Ties Haarman** Debye Institute for Nanomaterials Science, Utrecht University, 3584 CG Utrecht, The Netherlands

Complete contact information is available at: https://pubs.acs.org/10.1021/acs.jpcc.2c01521

Notes

The authors declare no competing financial interest.

ACKNOWLEDGMENTS

We gratefully acknowledge the synchrotron light source KARA and the KNMF at Karlsruhe, Germany, and the Taiwan Light Source at Hsinchu, Taiwan, for the provision of beam time. The authors thank the technical staff for their help with the XAS and RIXS measurements. The experiments were supported by an ERC Advanced Grant (Grant Agreement No. 340279-XRAYonACTIVE). A.H. was supported by JSPS KAKENHI Grant Number 21K13884. D.J.H. was supported by the Ministry of Science and Technology of Taiwan under Grant No. 106-2112-M-213-008-MY3. R.P.W. acknowledges funding by the German Ministry of Education and Research (BMBF), Grant No. 05K19GU2.

REFERENCES

(1) Picard, J. P.; Baud, G.; Besse, J. P.; Chevalier, R. Croissance Cristalline et étude Structurale de Co_3O_4 . J. Less Common. Met **1980**, 75, 99–104.

(2) Sparks, T. D.; Gurlo, A.; Gaultois, M. W.; Clarke, D. R. Revised Model for Thermopower and Site Inversion in Co_3O_4 Spinel. *Phys. Rev. B* **2018**, *98*, 024108.

(3) Roth, W. L. The Magnetic Structure of Co_3O_4 . J. Phys. Chem. Solids 1964, 25, 1–10.

(4) Mironova, N.; Skvortsova, V.; Ulmanis, U. Near Infrared Absorption Spectra in Co_3O_4 . Solid State Commun. **1994**, 91, 731–733.

(5) Goodenough, J. B.; Loeb, A. L. Theory of Ionic Ordering, Crystal Distortion, and Magnetic Exchange due to Covalent Forces in Spinels. *Phys. Rev.* **1955**, *98*, 391–408.

(6) Martens, J. W. D.; Peeters, W. L.; van Noort, H. M.; Erman, M. Optical Magneto-optical and Mössbauer Spectroscopy on Co^{3+} Substituted Cobalt Ferrite $Co^{2+}Fe_{2-x}Co^{3+}{}_{x}O_{4}$ (0 < x < 2). J. Phys. Chem. Solids **1985**, 46, 411–416.

(7) Vaz, C. A. F.; Altman, E. I.; Henrich, V. E. Exchange Bias and Interface Electronic Structure in $Ni/Co_3O_4(011)$. *Phys. Rev. B* 2010, *81*, 104428.

(8) Poizot, P.; Laruelle, S.; Grugeon, S.; Dupont, L.; Tarascon, J.-M. Nanosized Transition-Metal Oxides as Negative-Electrode Materials for Lithium-ion Batteries. *Nature* **2000**, *407*, 496–499.

(9) Du, N.; Zhang, H.; Chen, B. D.; Wu, J. B.; Ma, X. Y.; Liu, Z. H.; Zhang, Y. Q.; Yang, D. R.; Huang, X. H.; Tu, J. P. Porous Co_3O_4 Nanotubes Derived From $Co_4(CO)_{12}$ Clusters on Carbon Nanotube Templates: A Highly Efficient Material For Li-Battery. *Applications. Adv. Mater.* **2007**, *19*, 4505.

(10) Kim, T. W.; Woo, M. A.; Regis, M.; Choi, K.-S. Electrochemical Synthesis of Spinel Type $ZnCo_2O_4$ Electrodes for Use as Oxygen Evolution Reaction Catalysts. *J. Phys. Chem. Lett.* **2014**, *5*, 2370–2374.

(11) Wang, H.-Y.; Hung, S.-F.; Chen, H.-Y.; Chan, T.-S.; Chen, H. M.; Liu, B. In Operando Identification of Geometrical-Site-Dependent Water Oxidation Activity of Spinel Co₃O₄. *J. Am. Chem. Soc.* **2016**, *138*, 36–39.

(12) Li, Y.; Shen, W. Morphology-Dependent Nanocatalysts: Rodshaped Oxides. *Chem. Soc. Rev.* **2014**, *43*, 1543–1574.

(13) Singh, R. N.; Mishra, D.; Anindita; Sinha, A. S. K.; Singh, A. Novel Electrocatalysts for Generating Oxygen from Alkaline Water Electrolysis. *Electrochem. Commun.* **2007**, *9*, 1369–1373.

(14) Esswein, A. J.; McMurdo, M. J.; Ross, P. N.; Bell, A. T.; Tilley, T. D. Size-Dependent Activity of Co_3O_4 Nanoparticle Anodes for Alkaline Water Electrolysis. *J. Phys. Chem. C* **2009**, *113*, 15068–15072.

(15) Jiao, F.; Frei, H. Nanostructured Cobalt Oxide Clusters in Mesoporous Silica as Efficient Oxygen-Evolving Catalysts. Angew. *Chem. Int. Ed* **2009**, *48*, 1841–1844.

(16) Xi, L.; Tran, P. D.; Chiam, S. Y.; Bassi, P. S.; Mak, W. F.; Mulmudi, H. K.; Batabyal, S. K.; Barber, J.; Loo, J. S. C.; Wong, L. H. Co_3O_4 -Decorated Hematite Nanorods As an Effective Photoanode for Solar Water Oxidation. *J. Phys. Chem. C* **2012**, *116* (26), 13884– 13889.

(17) Zhang, X.; Chen, Y.-S.; Kamat, P. V.; Ptasinska, S. Probing Interfacial Electrochemistry on a Co_3O_4 Water Oxidation Catalyst Using Lab-Based Ambient Pressure X-ray Photoelectron Spectroscopy. J. Phys. Chem. C 2018, 122 (25), 13894–13901.

(18) Qiao, L.; Xiao, H. Y.; Meyer, H. M.; Sun, J. N.; Rouleau, C. M.; Puretzky, A. A.; Geohegan, D. B.; Ivanov, I. N.; Yoon, M.; Weber, W. J.; Biegalski, M. D. Nature of the Band Gap and Origin of the Electro-/Photo-activity of Co_3O_4 . J. Mater. Chem. C **2013**, 1, 4628– 4633.

(19) Singh, V.; Kosa, M.; Majhi, K.; Major, D. T. Putting DFT to the Test: A First-Principles Study of Electronic, Magnetic, and Optical Properties of Co_3O_4 . J. Chem. Theory Comput **2015**, 11 (1), 64–72. (20) Smart, T. J.; Pham, T. A.; Ping, Y.; Ogitsu, T. Optical Absorption Induced by Small Polaron Formation in Transition Metal Oxides: The Case of Co_3O_4 . Phys. Rev. Materials **2019**, 3, No. 102401. (21) Belova, I. D.; Roginskaya, Yu. E.; Shifrina, R. R.; Gagarin, S. G.; Plekhanov, Yu. V.; Venevtsev, Yu. N. Co (III) Ions High-spin Configuration in Nonstoichiometric Co_3O_4 Films. Solid State Commun. **1983**, 47, 577–584.

(22) Cook, J. G.; van der Meer, M. P. The Optical Properties of Sputtered Co_3O_4 Films. *Thin Solid Films* **1986**, 144, 165–176.

(23) Miedzinska, K. M. E.; Hollebone, B. R.; Cook, J. G. An Assigment of the Optical Absorption Spectrum of Mixed Valence Co_3O_4 Spinel Films. *J. Phys. Chem. Solids* **198**7, *48*, 649–656.

(24) Wang, X.; Chen, X.; Gao, L.; Zheng, H.; Zhang, Z.; Qian, Y. One-Dimensional Arrays of Co_3O_4 Nanoparticles: Synthesis, Characterization, and Optical and Electrochemical Properties. *J. Phys. Chem. B* **2004**, *108*, 16401–16404.

(25) Lima, A. F. Interpretation of the Optical Absorption Spectrum of Co_3O_4 with Normal Spinel Structure from First Principles Calculations. J. Phys. Chem. Solids **2014**, 75, 148–152.

(26) Wood, D. L.; Remeika, J. P. Optical Absorption of Tetrahedral Co³⁺ and Co²⁺ in Garnets. *J. Chem. Phys.* **1967**, *46*, 3595.

(27) van Schooneveld, M. M.; Gosselink, R. W.; Eggenhuisen, T. M.; Samarai, M. A.; Monney, C.; Zhou, K. J.; Schmitt, T.; de Groot, F. M. F. A Multispectroscopic Study of 3d Orbitals in Cobalt Carboxylates: The High Sensitivity of 2p3d Resonant X-ray Emission Spectroscopy to the Ligand Field. *Angew. Chem. Int. Ed* **2013**, *52*, 1170–1174.

(28) Hibberd, A. M.; Doan, H. Q.; Glass, E. N.; de Groot, F. M. F.; Hill, C. L.; Cuk, T. Co Polyoxometalates and a Co_3O_4 Thin Film Investigated by L-Edge X-ray Absorption Spectroscopy. *J. Phys. Chem.* C **2015**, *119*, 4173–4179.

(29) Elnaggar, H.; Wang, R.; Lafuerza, S.; Paris, E.; Komarek, A. C.; Guo, H.; Tseng, Y.; McNally, D.; Frati, F.; Haverkort, M. W.; Sikora, M.; Schmitt, T.; de Groot, F. M. F. Possible Absence of Trimeron Correlations Above the Verwey Temperature in Fe₃O₄. *Phys. Rev. B* **2020**, *101*, 085107.

(30) Winder, M.; Hariki, A.; Kuneš, J. X-ray Spectroscopy of the Rare-Earth Nickelate LuNiO₃: LDA+DMFT Study. *Phys. Rev. B* **2020**, *102*, 085155.

(31) Lu, Y.; Betto, D.; Fursich, K.; Suzuki, H.; Kim, H.-H.; Cristiani, G.; Logvenov, G.; Brookes, N. B.; Benckiser, E.; Haverkort, M. W.; Khaliullin, G.; Le Tacon, M.; Minola, M.; Keimer, B. Site-Selective Probe of Magnetic Excitations in Rare-Earth Nickelates Using Resonant Inelastic X-ray Scattering. *Phys. Rev. X* **2018**, *8*, 031014.

(32) Huang, H. Y.; Chen, Z. Y.; Wang, R.-P.; de Groot, F. M. F.; Wu, W. B.; Okamoto, J.; Chainani, A.; Singh, A.; Li, Z.-Y.; Zhou, J. S.; et al. Jahn-Teller Distortion Driven Magnetic Polarons in Magnetite. *Nat. Commun.* **2017**, *8*, 15929.

(33) Magnuson, M.; Butorin, S.; Guo, J.-H.; Nordgren, J. Electronic Structure Investigation of CoO by Means of Soft X-ray Scattering. *Phys. Rev. B* **2002**, *65*, 205106.

(34) Hariki, A.; Winder, M.; Kuneš, J. Continuum Charge Excitations in High-Valence Transition-Metal Oxides Revealed by Resonant Inelastic X-Ray Scattering. *Phys. Rev. Lett.* **2018**, *121*, 126403.

(35) Elnaggar, H.; Wang, R.-P.; Lafuerza, S.; Paris, E.; Tseng, Y.; McNally, D.; Komarek, A.; Haverkort, M.; Sikora, M.; Schmitt, T.; de Groot, F. M. F. Magnetic Contrast at Spin-Flip Excitations: An Advanced X-Ray Spectroscopy Tool to Study Magnetic-Ordering. *ACS Appl. Mater. Interfaces* **2019**, *11*, 36213–36220.

(36) Wang, R.-P.; Liu, B.; Green, R. J.; Delgado-Jaime, M. U.; Ghiasi, M.; Schmitt, T.; van Schooneveld, M. M.; de Groot, F. M. F. Charge-Transfer Analysis of 2p3d Resonant Inelastic X-ray Scattering of Cobalt Sulfide and Halides. *J. Phys. Chem. C* **2017**, *121* (45), 24919–24928.

(37) Al Samarai, M.; Delgado-Jaime, M. U.; Ishii, H.; Hiraoka, N.; Tsuei, K.-D.; Rueff, J.–P.; Lassale-Kaiser, B.; Weckhuysen, B. M.; de Groot, F. M. F. 1s3p Resonant Inelastic X-ray Scattering of Cobalt Oxides and Sulfides. *J. Phys. Chem. C* **2016**, *120*, 24063–24069.

(38) Kwon, G.; Jang, H.; Lee, J.-S.; Mane, A.; Mandia, D. J.; Soltau, S. R.; Utschig, L. M.; Martinson, A. B. F.; Tiede, D. M.; Kim, H.; Kim, J. Resolution of Electronic and Structural Factors Underlying Oxygen-Evolving Performance in Amorphous Cobalt Oxide Catalysts. *J. Am. Chem. Soc.* **2018**, *140*, 10710–10720.

(39) Lai, C. H.; Fung, H. S.; Wu, W. B.; Huang, H. Y.; Fu, H. W.; Lin, S. W.; Huang, S. W.; Chiu, C. C.; Wang, D. J.; Huang, L. J.; et al. Highly Efficient Beamline and Spectrometer for Inelastic Soft X-ray Scattering at High Resolution. *J. Synchrotron Rad* **2014**, *21*, 325–332. (40) Fung, H. S.; Chen, C. T.; Huang, L. J.; Chang, C. H.; Chung, S. C.; Wang, D. J.; Tseng, T. C.; Tsang, K. L. A Novel Active Grating Monochromator — Active Grating Spectrometer Beamline System for Inelastic Soft-X-ray Scattering Experiments. *AIP Conf. Proc.* 2003, 705, 655.

(41) de Groot, F. M. F. Multiplet Effects in X-ray Spectroscopy. Coord. Chem. Rev. 2005, 249, 31–63.

(42) Haverkort, M. W.; Zwierzycki, M.; Andersen, O. K. Multiplet Ligand-Field Theory Using Wannier Orbitals. *Phys. Rev. B* 2012, *85*, 165113.

(43) Tomiyasu, K.; Okamoto, J.; Huang, H. Y.; Chen, Z. Y.; Sinaga, E. P.; Wu, W. B.; Chu, Y. Y.; Singh, A.; Wang, R.-P.; de Groot, F. M. F.; et al. Coulomb Correlations Intertwined with Spin and Orbital Excitations in LaCoO₃. *Phys. Rev. Lett.* **2017**, *119*, 196402.

(44) Liu, B.; Wang, R.-P.; Glass, E. N.; Hill, C. L.; Cuk, T.; Okamoto, J.; Huang, D. J.; van Schooneveld, M. M.; de Groot, F. M. F. Distorted Tetrahedral Co^{II} in $K_5H[CoW_{12}O_{40}]\cdot xH_2O$ Probed by 2p3d Resonant Inelastic X-ray Scattering. *Inorg. Chem.* **2016**, *55* (20), 10152–10160.

(45) Wang, R.-P.; Geessinck, J.; Elnaggar, H.; Birkholzer, Y. A.; Tomiyasu, K.; Okamoto, J.; Liu, B.; Du, C.-H.; Huang, D.-J.; Koster, G.; de Groot, F. M. F. Low-Energy Orbital Excitations in Strained $LaCoO_3$ films. *Phys. Rev. B* **2019**, *100*, 165148.

(46) Cramer, S. P.; de Groot, F. M. F.; Ma, Y.; Chen, C. T.; Sette, F.; Kipke, C. A.; Eichhorn, D. M.; Chan, M. K.; Armstrong, W. H.; Libby, E.; et al. Ligand Field Strengths and Oxidation States from Manganese L-edge Spectroscopy. J. Am. Chem. Soc. **1991**, 113, 7937–7940.

(47) Hariki, A.; Winder, M.; Uozumi, T.; Kuneš, J. LDA+DMFT Approach to Resonant Inelastic X-ray Scattering in Correlated Materials. *Phys. Rev. B* **2020**, *101*, 115130.

(48) Achkar, A. J.; Regier, T. Z.; Wadati, H.; Kim, Y.-J.; Zhang, H.; Hawthorn, D. G. Bulk Sensitive X-ray Absorption Spectroscopy Free of Selfabsorption Effects. *Phys. Rev. B* **2011**, *83*, 081106.

(49) Wang, R.-P.; Elnaggar, H.; Titus, C. J.; Tomiyasu, K.; Geessinck, J.; Koster, G.; Frati, F.; Okamoto, J.; Huang, D.-J.; de Groot, F. M. F. Saturation and Self-absorption Effects in the Angle Dependent 2p3d Resonant Inelastic X-ray Scattering Spectra of Co³⁺. *J. Synchrotron Rad* **2020**, *27*, 979–987.

(50) Chen, J.; Wu, X.; Selloni, A. Electronic Structure and Bonding Properties of Cobalt Oxide in the Spinel Structure. *Phys. Rev. B* 2011, 83, 245204.

Recommended by ACS

RAIRS Characterization of CO and O Coadsorption on Cu(111)

Diyu Zhang, Ludo B. F. Juurlink, *et al.* JULY 28, 2022 THE JOURNAL OF PHYSICAL CHEMISTRY C

pubs.acs.org/JPCC

READ 🗹

From the Free Ligand to the Transition Metal Complex: FeEDTA⁻ Formation Seen at Ligand K-Edges

Sebastian Eckert, Alexander Föhlisch, et al.

JUNE 28, 2022 INORGANIC CHEMISTRY

READ 🗹

CO Adsorbate Promotes Polaron Photoactivity on the Reduced Rutile TiO₂(110) Surface

Cheng Cheng, Oleg V. Prezhdo, *et al.* DECEMBER 30, 2021 JACS AU

READ 🗹

Chemical Information in the L_3 X-ray Absorption Spectra of Molybdenum Compounds by High-Energy-Resolution Detection and Density Functional Theory

Artem Svyazhin, Pieter Glatzel, *et al.* DECEMBER 27, 2021 INORGANIC CHEMISTRY

READ 🗹

Get More Suggestions >