

Femtosecond Charge Density Modulations in Photoexcited CuWO₄

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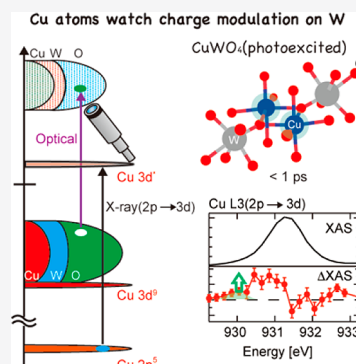
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ABSTRACT: Copper tungstate (CuWO₄) is an important semiconductor with a sophisticated and debatable electronic structure that has a direct impact on its chemistry. Using the PAL-XFEL source, we study the electronic dynamics of photoexcited CuWO₄. The Cu L₃ X-ray absorption spectrum shifts to lower energy upon photoexcitation, which implies that the photoexcitation process from the oxygen valence band to the tungsten conduction band effectively increases the charge density on the Cu atoms. The decay time of this spectral change is 400 fs indicating that the increased charge density exists only for a very short time and relaxes electronically. The initial increased charge density gives rise to a structural change on a time scale longer than 200 ps.



1. INTRODUCTION

Copper tungstate (CuWO₄) is an important functional material that has gained a lot of attention in materials science because of its small optical band gap and electrical conductivity. In particular, it is a promising candidate for photocatalysis¹ and a potential photoelectrode for the water-splitting reaction.^{2–6} CuWO₄ has a triclinic distorted wolframite-type structure due to the Jahn–Teller distortion around the Cu²⁺ ions.⁷ It is known as an n-type semiconductor and its valence band is believed to be composed mainly of oxygen 2p states that are hybridized with tungsten and copper s, p, and d states.^{8,9} The conduction band contains mainly W 5d states hybridized with the oxygen 2p states. In addition, it has been claimed that the single Cu 3d hole contributes to the bottom of the conduction band, which shifts the conduction band minimum in CuWO₄ downward compared to WO₃.^{10–14} However, the origin of charge carriers and their behavior in CuWO₄ are not well understood due to the scarcity of the experimental studies that probe the relaxation dynamics of the optically excited Cu electronic states, particularly in the fs/ps time domain.^{15,16} Therefore, we believe that understanding the ultrafast electronic transitions in CuWO₄ in this time domain could provide a clearer insight on the nature of the electronic structure of CuWO₄, which is highly debatable in literature.^{8,10,11,14,17}

In this work, we conducted the first investigation of the ultrafast charge carrier dynamics in CuWO₄ using time-resolved optical and X-ray spectroscopies in the 100 fs to 200 ps time scale with a focus on the role of the copper 3d states in the charge

carrier transport process. Time-resolved X-ray absorption spectroscopy (XAS) provides detailed information on the ultrafast charge carrier dynamics and recombination in several metal oxide semiconductors.^{18–20} Charge carrier dynamics of water splitting semiconductors (TiO₂,^{21–23} α-Fe₂O₃,^{24–27} BiVO₄,^{28,29} WO₃,^{30–32} ZnO,³³ and CsPbBr₃ and CsPb(ClBr)₃ perovskites³⁴) were successfully observed using ultrafast XAS. Herein, we present transient Cu L₃ XAS studies on the photoexcited state of CuWO₄. Cu L₃ XAS is attributed to electron transition from Cu 2p orbitals to Cu 3d orbitals, which reflects the chemical state of the Cu atoms directly. The optical laser excitation triggers a transition from the oxygen 2p valence band to the W 5d conduction band.⁸ The empty Cu 3d state is not active in the optical transition and, as such, plays the role of a spectator state. During the X-ray excitation process, an electron is excited to the Cu 3d band and one can observe its reaction to the optical excitation. This role as a spectator state makes it attractive to study its response to the optical excitation, in order to determine the model of the electronic structure and the dynamics in electronic states of CuWO₄.

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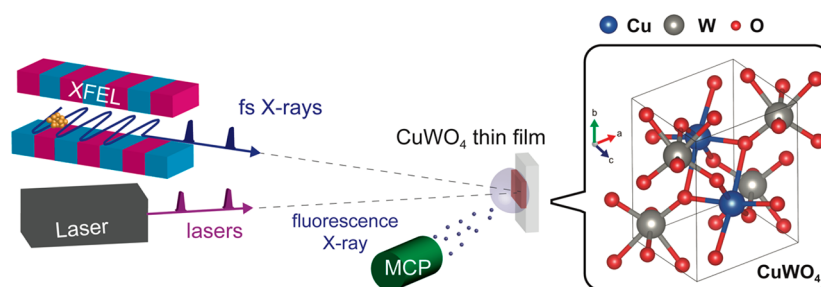


Figure 1. Time-resolved X-ray absorption spectroscopy (XAS) on CuWO_4 : experimental setup of the pump–probe XAS in PAL-XFEL and the crystal structure of CuWO_4 .

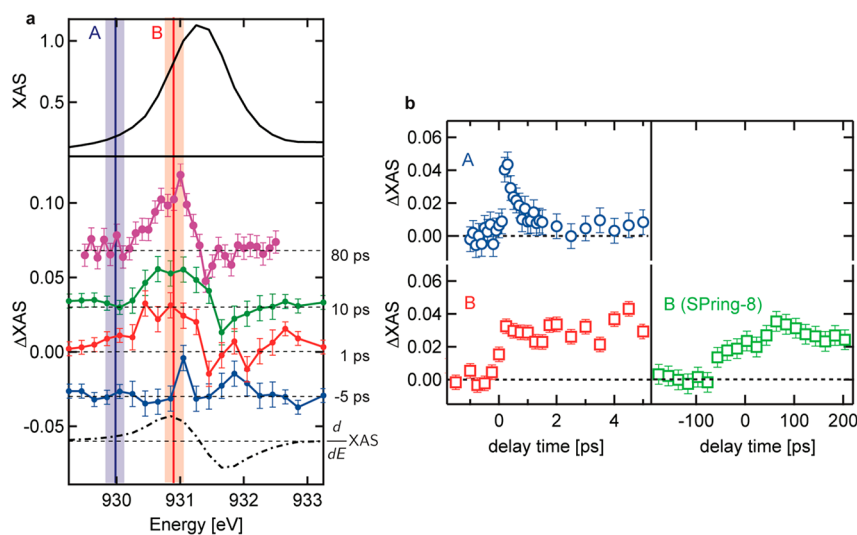


Figure 2. (a) Normalized Cu L_3 XAS (top panel) and the transient X-ray absorption spectroscopy (XAS) data of Cu L_3 edge in CuWO_4 at several delay times (-5 , 1 , 10 , and 80 ps with respect to -10 ps) (bottom panel). The transient XAS spectra were normalized according to the reference Cu L_3 XAS. (b) Kinetic traces of the Cu L_3 edge at peaks A and B measured at PAL-XFEL and the kinetic trace of Cu L_3 edge XAS at peak B for longer delay times measured at SPring-8. The intensities were normalized according to the reference Cu L_3 XAS. The error bars for the transient XAS were estimated from the standard deviation of the transient XAS at a negative delay time. Due to shot-by-shot variations of the X-ray intensity, there remains a point-by-point error that can be larger than the standard deviation. The error bars for the kinetic traces were estimated from the standard deviation of the unpumped XAS intensity.

2. EXPERIMENTS

The CuWO_4 sample was a polycrystalline thin film with a thickness of 50 nm synthesized using pulsed laser deposition (PLD) supported on a conductive glass substrate (more details of the sample preparation and physical characterizations are described in the [Supporting Information](#)). Pump–probe Cu L_3 XAS experiments were conducted in the soft X-ray spectroscopy and scattering (SSS) beamline at PAL-XFEL.^{35–37} A simple illustration of the experimental setup is shown in [Figure 1](#). The sample was placed at the focal point of the X-rays (the focal size of the X-rays was less than $50\ \mu\text{m}$ (H) $\times$ $50\ \mu\text{m}$ (V)). The 400 nm optical pump laser was transferred to the focal position of the X-rays. The focal size of the optical laser was $\sim 210\ \mu\text{m}$ (H) $\times$ $210\ \mu\text{m}$ (V). XAS signals were obtained using a microchannel plate (MCP) detector. A more detailed experimental setup is shown in [Figure S3](#). For the observation of transient XAS at longer delay time (>80 ps), pump–probe experiments were conducted at BL07LSU, SPring-8.³⁸ The detailed experimental setup at SPring-8 as well as other experimental details (optical pump–probe experiments and static Cu K-edge XAS) are described in the [Supporting Information](#).

The X-ray intensity fluctuates shot-by-shot in PAL-XFEL, and some X-ray shots have too large or too small intensities to obtain

XAS signals. Therefore, such X-ray shots were removed from data sets since they worsen the signal to noise ratio (S/N). Details of analyzing the X-ray data to calculate XAS are described in [Supporting Information](#). In addition to the shot-by-shot fluctuations, variations in the intensity of XFEL pulses were observed over several hours, which might affect the XAS signal and result in artificial differential spectra. To avoid these artifacts, the XAS at each delay time and those at the reference were measured in sequence. That is to say, a reference XAS spectrum at a delay time of -10 ps, i.e., an excitation laser pulse arrives at the sample 10 ps after an X-ray pulse arrives, was measured for each measurement in order to calculate a difference spectrum for each delay time. Each difference XAS spectrum (ΔXAS) at a delay (Δt) was calculated by subtracting XAS at -10 ps from XAS at Δt , i.e.,

$$\Delta\text{XAS} = \text{XAS}(\Delta t [\text{ps}]) - \text{XAS}(\Delta t = -10 \text{ ps})$$

At SPring-8, the laser pump repetition rate was set to half of the X-ray probe repetition rate. XAS signals with or without laser shots were accumulated consecutively.

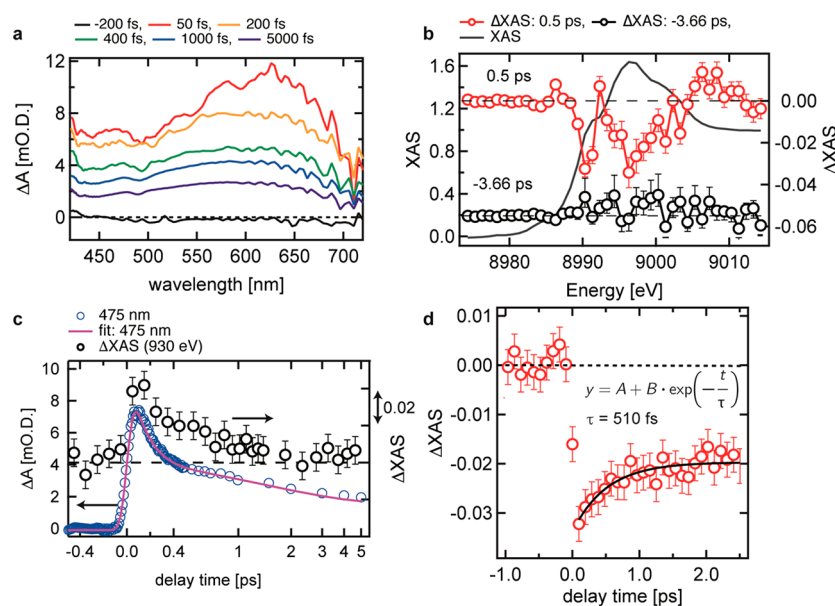


Figure 3. (a) Transient optical absorbance of CuWO_4 from 420 to 720 nm. (b) Transient Cu K-edge XAS of CuWO_4 before photoexcitation (-3.66 ps) and after photoexcitation (0.5 ps). The difference spectra were normalized according to the Cu K-edge XAS in the optical ground state. (c) Kinetic trace of the transient optical absorbance of CuWO_4 at 475 nm. (d) Kinetic trace of transient Cu K-edge XAS of CuWO_4 at 8980 eV. The intensity was normalized according to the difference XAS shown in part b.

3. RESULTS

3.1. Transient Cu L_3 XAS of CuWO_4 . The transient XAS spectra at several delays are displayed in Figure 2. The difference XAS spectra at delays of -5 ps, 1 ps, and 10 ps were measured at PAL-XFEL, whereas the difference XAS spectrum at a delay of 80 ps was measured at SPring-8. After optical excitation, one observes an absorption increase at energies below the static L_3 edge maximum of 931.4 eV and in addition there is a minor absorption decrease at energies above the L_3 edge maximum. We note that the observed spectral changes do not agree with the first derivative of Cu L_3 XAS (dashed line). Figure 2b (right panels) shows the kinetic traces of the L_3 edge at 930.0 eV (position A) and 930.8 eV (position B). The kinetic trace at position B shows a rise related to the time resolution of the experiment followed by a constant intensity increase over the whole range of 5 ps. The absorption change at position B was measured up to 200 ps, but at a delay time of 200 ps, and was not recovered to its original ground state value at this time delay (Figure 2b, SPring-8). On the other hand, the kinetic trace of peak A (930.0 eV) decays on a much shorter time scale, where the kinetic constant of this fast process was found to be 400 ± 160 fs (Figure S4).

3.2. Pump–Probe Optical Spectroscopy and Transient Cu K-Edge XAS of CuWO_4 . A similar fast decay process was also observed by optical pump–probe spectroscopy shown in Figure 3a and Figure 3c and by pump–probe Cu K-edge XAS conducted at SPring-8 angstrom compact free electron laser (SACLA)^{39–43} shown in Figure 3b and Figure 3d. In the optical spectroscopy study, a broad transient absorption was observed between 420 and 720 nm. This broad absorption should be mainly attributed to excited electrons distributed in the conduction band. Since excited electrons have different energies and transition to different final states, a broad absorption can be observed after photoexcitation. As described in Figure S2, there were four time constants extracted from a rate-equation model. The Cu K-edge XAS after photoexcitation is interpreted by the shift of the Cu K-edge XAS to lower energy. This result agrees

with the increase of the electron density at the Cu sites. A fast kinetic process was observed at 8990.3 eV, which decays by a delay of 2 ps. The kinetic constant of this process was estimated to be 510 ± 150 fs. This value is consistent with the kinetic constant estimated by Cu L_3 edge within the error limits. This excess decrease of the X-ray intensity could reflect the excess charge density at the Cu atoms in agreement with the Cu L_3 edge data.

3.3. Interpretation of the Transient Cu L_3 XAS. We first note that the peak position of Cu^{2+} in the Cu L_3 XAS is lower than that of both Cu^{1+} and Cu^{3+} (for example, refs 44–46), which is in contrast to all other transition metal ions. The reason is that the ground state of Cu^{2+} is $3d^9$ and the only allowed final state is $2p^5 3d^{10}$; hence there is only a single peak observed at the L_3 edge. The peak position of Cu^{3+} appears at higher energy position due to the larger effective charge of the copper ion. Also, the first L_3 peak of Cu^{1+} appears at higher energy since the $3d$ band of Cu^{1+} is full and the intensity is given by the transition to a $4s$ orbital that forms an exciton.⁴⁷ Interestingly neither a change to Cu^{1+} or Cu^{3+} will allow a peak at an energy lower than Cu^{2+} . In other words, the spectral change at 930.0 eV cannot be assigned to change of the valence state of the copper ion.

Instead, we assign this feature to Cu^{2+} that has a modified local electronic structure. The optical excitation moves an electron from the oxygen $2p$ valence band to the tungsten $5d$ conduction band, while Cu^{2+} is essentially a spectator ion. Since the charge density of the copper s and p states is larger in the W $5d$ conduction band than in the oxygen $2p$ valence band, the effective charge density on the Cu^{2+} ions will increase creating a $\text{Cu}^{2+} 3d^9$ ion that has enhanced electron density, which will shift the L_3 edge to lower energy. This enhanced charge density is contained within band states with copper s and p character, and being a delocalized band state, this extra charge will rapidly delocalize, faster than the time constant of 400 fs. The change in charge distribution can affect the vibrational potential in the excited system due to electron–lattice coupling.⁴⁸ Under such situation, low-frequency modes move toward this new origin

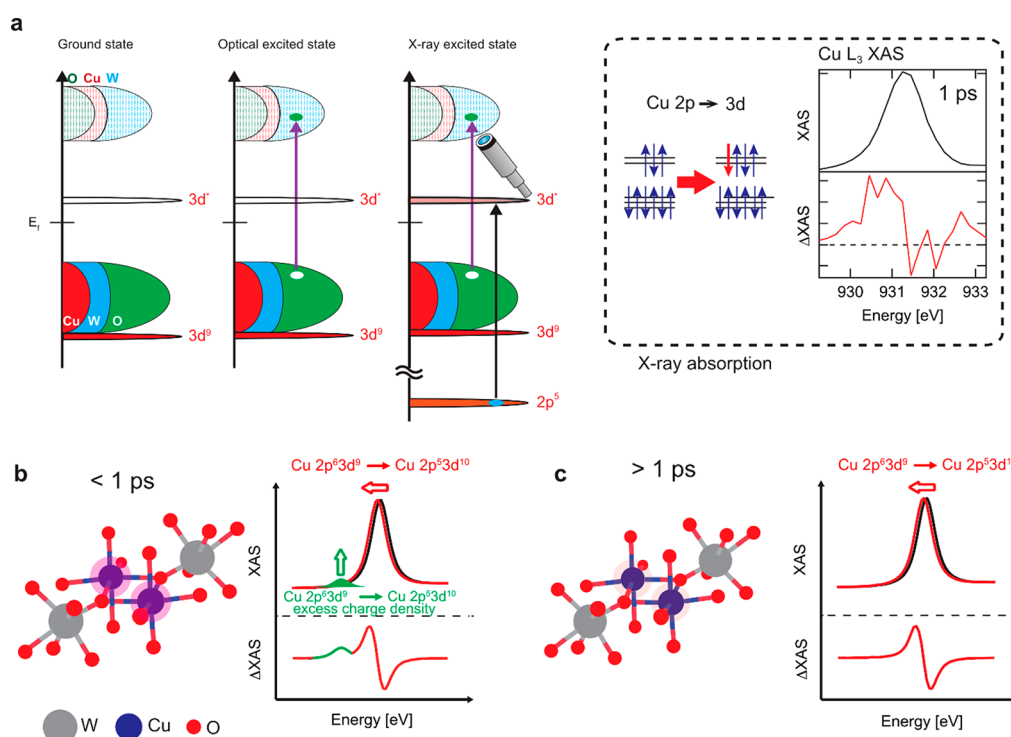


Figure 4. (a) A sketch that explains the optical and X-ray excitation processes in CuWO₄ and the Cu L₃-edge transient XAS spectra at a delay time of 10 ps. Illustrations of the charge density at Cu sites in CuWO₄ after photoexcitation below a delay of 1 ps (b) and above a delay of 1 ps (c).

with the modified interatomic distances and made additional modification to charge distribution. The experiment thus uses the copper L₃ edge spectrum as an indicator of the local charge density changes (and phonon modes) on the copper site following optical laser excitation.

3.4. Photoexcitation Dynamics of CuWO₄. Two main changes were observed by transient Cu L₃ XAS upon optical excitation of CuWO₄: (i) An initial photoexcited state is created by optical excitation, creating enhanced charge density at the copper sites, decaying with a time constant of 400 fs. The enhanced charge density is related to copper s and p states, and the valence remains Cu²⁺. (ii) The photoexcited state undergoes a decay process to form a metastable state that has a lifetime of more than 200 ps. The metastable state has increased charge density with respect to the optical ground state but reduced with respect to the initial excited state. This metastable state must involve structural reorganization of atoms related to a polaronic state while the valence remains Cu²⁺. It should be noted that local structural changes can also shift the Cu L₃ edge XAS to lower energy; i.e., if the Cu–O distance is elongated by 10%, calculations show that the Cu L₃ XAS shifts by ~0.2 eV.⁴⁹ The local structural change affects more directly the spectral features of the Cu K-edge XAS in higher energy, which is visible at ~9000 eV at delays of 2.3 and 8.3 ps (see Figure S5). Local structural changes are related to the formation of polaronic states. Regarding the polaron state formation in CuWO₄, Hoang et al. suggested that an electron hole should be more localized on the O atoms while an electron would be more localized on metal sites.¹⁴ Considering their results, we conclude that the effective negative charge on the O atoms would be decreased by + δ while the effective positive charge on W atoms and Cu atoms would also decrease by $-\delta$, reducing the charge difference and thereby increasing the bond length.

4. DISCUSSION

4.1. Model of the Electronic Structure of CuWO₄. As mentioned above, the electronic structure of CuWO₄ is still an open question and there have been models suggested by several authors. For instance, Tian et al.⁸ reported that the bottom of the conduction band of CuWO₄ comes from Cu 3d orbitals while the top of the valence band is formed by Cu 3d orbitals and O 2p orbitals. Hoang et al.¹⁴ pointed out that O 2p orbitals are dominant in the valence band of CuWO₄ while its conduction band is formed mainly by Cu 3d orbitals and W 5d orbitals. Khyzhun et al.¹¹ suggested that the conduction band of CuWO₄ is a W 5d-like state while its valence band is an O 2p-like state and they also assume the existence of Cu³⁺.¹⁰ Considering the previous reports and our transient XAS results, we assume a combined electronic structure of WO₃-type structure with an oxygen 2p valence band and a W 5d conduction band. The 3d-electrons of the Cu²⁺ ion contribute to the oxygen 2p valence band, while its occupied 3d electrons are positioned at lower energies, in line with all other Cu²⁺ oxides that are charge transfer insulators.¹⁷

In order to discuss the electronic structure of CuWO₄ in detail, we first introduce the electronic structure of the binary parent oxides WO₃ and CuO. WO₃ is a semiconductor with a valence band mainly composed of oxygen 2p character. The oxygen 2p state hybridizes strongly with the tungsten 5d, 6s, and 6p states that yield a considerable amount of W character in the valence band. The conduction band consists of the antibonding combinations of oxygen 2p states with tungsten 5d, 6s, and 6p states. For simplicity, this could be called the tungsten 5d band, but it also contains significant contributions of tungsten s and p character. The electronic structure of CuO is described as a charge transfer insulator,¹⁷ where the Cu 3d states are split between the lower Hubbard band below the oxygen 2p band and the empty upper Hubbard band that has one hole state.⁵⁰

CuWO₄ can be considered as a combination of both oxides. The (top of the) valence band contains mainly oxygen 2p character mixed with W 5d and sp states plus Cu 3d and sp states. The conduction band is dominated by W 5d but also contains oxygen 2p and copper 3d and sp states. In between, the upper Hubbard band of Cu 3d sits in the band gap (see Figure 4a).

4.2. The 400 fs Metastable Charge Density Modulation. The laser excitation of CuWO₄ involves a transition from the oxygen 2p valence band to the tungsten 5d conduction band. This creates a ligand hole plus an electron in the W 5d band. The Cu 3d states are spectators to this laser excitation since the Cu 3d states are not influenced by the photoexcitation as described (section 3.3). The electronic structure of the copper sites is however affected because the excited electron in the conduction band has partly copper 4s/4p character. This additional charge at the Cu site will decrease the binding energy of the 2p core states, following the general rule that the lower is the number of electrons at a metal site (i.e., a higher valency) the higher is its binding energy. Thus, the laser excitation creates a “spectator” copper site with a lower binding energy of its 2p core states.

The Cu 2p XAS spectrum of CuWO₄ shows essentially one sharp peak and at higher energy some weaker intensity. The sharp peak is related to the 3d⁹ → 2p⁵3d¹⁰ transition. The laser excited CuWO₄ sample will also have a single 3d⁹ → 2p⁵3d¹⁰ transition, but due to the reduced binding energy of the 2p core state, this transition will be shifted to lower excitation energy. The small absorption appears due to this shifted 3d⁹ → 2p⁵3d¹⁰ transition (the green small peak in Figure 4b). In other words, we can track the charge density change of the Cu spectator state by measuring the time evolution of the transient signal at an excitation energy below the copper L₃ edge (see Figure 4b and Figure 4c). The Cu state with the modified charge density has a lifetime of only 400 fs. The 400 nm laser excites the electrons ~3.1 eV up in energy, and with a band gap of 2.3 eV they are excited in the bottom 0.8 eV of the conduction band. The corresponding modified charge density can lose energy via carrier cooling and electronic decay channels to the lowest states of the conduction band. Potentially this electronic picture could be modified by phonons affecting the charge density on copper via a changing Cu–O distance.

4.3. Effect on Photocatalysis. The observations above provide insights on the photocatalytic behavior of CuWO₄ compared to WO₃ and CuO. Adding Cu to WO₃ modifies the band gap of the system, making it capable of absorbing optical light in the visible range of the solar spectrum which enhances its photocatalytic performance. The system still requires a large overpotential to drive photocatalytic processes such as water splitting. This is possibly due to the formation of a polaronic state. The initial optical excitation creates a modified charge density that is viewed from the Cu site in our study. After the decay of the original optical excitation within 400 fs, the system does not return to its original state, which is a clear indication of the formation of a polaronic state. It is believed that polarons are formed due to lattice distortion around the light-absorbing atom, resulting in trapping and hence reducing the mobility of electrons, which effects the photocatalytic efficiency of the system. The formation of this polaronic state has been observed in several metal oxide systems.^{26,51,52}

Furthermore, it is important to note that in order to create an effective photocatalyst, one needs to optimize the system of photon capture, which is mainly governed by the band gap. The latter is determined by the overlap between tungsten and oxygen. Copper remains Cu²⁺ throughout the whole process of

the optical excitation and the 400 fs decay. In that sense the Cu state is described as a spectator that is not redox active but rather creates a better (electronic and structural) environment for the WO₃ system that receives the optical photon.

CONCLUDING REMARKS

In summary, by tracking a 50 fs, 3.1 eV optical laser excitation in CuWO₄ via a pump–probe delayed X-ray excitation 1.0 eV before the Cu L₃ edge, we have identified an increased charge density at the Cu site that has a decay time of 400 fs. This fast decay implies that the modified charge density decays electronically via carrier cooling to a longer-lived electronic excited state. It would be very interesting to investigate if this type of charge modulation with decay times of a few hundred femtoseconds is a general phenomenon in transition metal oxides and if this phenomenon is correlated with photocatalytic activity that can be modified by sample preparation and by the parameters in the laser excitation.

ASSOCIATED CONTENT

Supporting Information

The Supporting Information is available free of charge at <https://pubs.acs.org/doi/10.1021/acs.jpcc.0c10525>.

Physical characterizations of the CuWO₄ thin films (SEM, XRD, XPS, AFM, UV–Vis spectrum, and a Tauc plot), details of the transient optical spectroscopy, experimental details of transient XAS spectroscopy (PAL-XFEL, SPring-8 and SACLA), and supplemental results of transient XAS at SACLA (PDF)

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Notes

The authors declare no competing financial interest.

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REFERENCES

- (1) Momeni, M. M. Fabrication of copper decorated tungsten oxide–titanium oxide nanotubes by photochemical deposition technique and their photocatalytic application under visible light. *Appl. Surf. Sci.* **2015**, 357, 160–166.
- (2) Lhermitte, C. R.; Bartlett, B. M. Advancing the Chemistry of CuWO₄ for Photoelectrochemical Water Oxidation. *Acc. Chem. Res.* **2016**, 49, 1121–1129.
- (3) Valenti, M.; Dolat, D.; Biskos, G.; Schmidt-Ott, A.; Smith, W. A. Enhancement of the Photoelectrochemical Performance of CuWO₄ Thin Films for Solar Water Splitting by Plasmonic Nanoparticle Functionalization. *J. Phys. Chem. C* **2015**, 119, 2096–2104.
- (4) Yourey, J. E.; Pyper, K. J.; Kurtz, J. B.; Bartlett, B. M. Chemical Stability of CuWO₄ for Photoelectrochemical Water Oxidation. *J. Phys. Chem. C* **2013**, 117, 8708–8718.
- (5) Hill, J. C.; Ping, Y.; Galli, G. A.; Choi, K.-S. Synthesis, photoelectrochemical properties, and first principles study of n-type CuW_{1-x}Mo_xO₄ electrodes showing enhanced visible light absorption. *Energy Environ. Sci.* **2013**, 6, 2440–2446.
- (6) Benko, F. A.; MacLaurin, C. L.; Koffyberg, F. P. CuWO₄ and Cu₃WO₆ as anodes for the photoelectrolysis of water. *Mater. Res. Bull.* **1982**, 17, 133–136.
- (7) Schofield, P. F.; Knight, K. S.; Redfern, S. A. T.; Cressey, G. Distortion Characteristics Across the Structural Phase Transition in (Cu_{1-x}Zn_x)WO₄. *Acta Crystallogr., Sect. B: Struct. Sci.* **1997**, 53, 102–112.
- (8) Tian, C. M.; Jiang, M.; Tang, D.; Qiao, L.; Xiao, H. Y.; Oropeza, F. E.; Hofmann, J. P.; Hensen, E. J. M.; Tadich, A.; Li, W.; et al. Elucidating the electronic structure of CuWO₄ thin films for enhanced photoelectrochemical water splitting. *J. Mater. Chem. A* **2019**, 7, 11895–11907.
- (9) Kihlberg, L.; Gebert, E. CuWO₄, a distorted Wolframite-type structure. *Acta Crystallogr., Sect. B: Struct. Crystallogr. Cryst. Chem.* **1970**, 26, 1020–1026.
- (10) Khyzhun, O. Y.; Strunskus, T.; Cramm, S.; Solonin, Y. M. Electronic structure of CuWO₄: XPS, XES and NEXAFS studies. *J. Alloys Compd.* **2005**, 389, 14–20.
- (11) Khyzhun, O. Y.; Bekenev, V. L.; Solonin, Y. M. First-principles calculations and X-ray spectroscopy studies of the electronic structure of CuWO₄. *J. Alloys Compd.* **2009**, 480, 184–189.
- (12) Lalić, M. V.; Popović, Z. S.; Vukajlović, F. R. Ab initio study of electronic, magnetic and optical properties of CuWO₄ tungstate. *Comput. Mater. Sci.* **2011**, 50, 1179–1186.
- (13) Gaillard, N.; Chang, Y.; Braun, A.; DeAngelis, A. Copper Tungstate (CuWO₄)-Based Materials for Photoelectrochemical Hydrogen Production. *MRS Proc.* **2012**, 1446, 31–36.
- (14) Hoang, K.; Oh, M.; Choi, Y. Electronic structure, polaron formation, and functional properties in transition-metal tungstates. *RSC Adv.* **2018**, 8, 4191–4196.

- (15) Hirst, J.; Müller, S.; Peeters, D.; Sadlo, A.; Mai, L.; Reyes, O. M.; Friedrich, D.; Mitoraj, D.; Devi, A.; Beranek, R.; et al. Comparative Study of Photocarrier Dynamics in CVD-deposited CuWO_4 , CuO , and WO_3 Thin Films for Photoelectrocatalysis. *Z. Phys. Chem. (Muenchen, Ger.)* **2020**, 234, 699–717.
- (16) Peeters, D.; Mendoza Reyes, O.; Mai, L.; Sadlo, A.; Cwik, S.; Rogalla, D.; Becker, H. W.; Schütz, H. M.; Hirst, J.; Müller, S.; et al. CVD-grown copper tungstate thin films for solar water splitting. *J. Mater. Chem. A* **2018**, 6, 10206–10216.
- (17) Zaanen, J.; Sawatzky, G. A.; Allen, J. W. Band gaps and electronic structure of transition-metal compounds. *Phys. Rev. Lett.* **1985**, 55, 418–421.
- (18) Kraus, P. M.; Zürich, M.; Cushing, S. K.; Neumark, D. M.; Leone, S. R. The ultrafast X-ray spectroscopic revolution in chemical dynamics. *Nat. Rev. Chem.* **2018**, 2, 82–94.
- (19) Chergui, M.; Collet, E. Photoinduced Structural Dynamics of Molecular Systems Mapped by Time-Resolved X-ray Methods. *Chem. Rev. (Washington, DC, U. S.)* **2017**, 117, 11025–11065.
- (20) Milne, C. J.; Penfold, T. J.; Chergui, M. Recent experimental and theoretical developments in time-resolved X-ray spectroscopies. *Coord. Chem. Rev.* **2014**, 277–278, 44–68.
- (21) Obara, Y.; Ito, H.; Ito, T.; Kurahashi, N.; Thürmer, S.; Tanaka, H.; Katayama, T.; Togashi, T.; Owada, S.; Yamamoto, Y.-i.; et al. Femtosecond time-resolved X-ray absorption spectroscopy of anatase TiO_2 nanoparticles using XFEL. *Struct. Dyn.* **2017**, 4, No. 044033.
- (22) Santomauro, F. G.; Lübcke, A.; Rittmann, J.; Baldini, E.; Ferrer, A.; Silatani, M.; Zimmermann, P.; Grübel, S.; Johnson, J. A.; Mariager, S. O.; et al. Femtosecond X-ray absorption study of electron localization in photoexcited anatase TiO_2 . *Sci. Rep.* **2015**, 5, 14834.
- (23) Rittmann-Frank, M. H.; Milne, C. J.; Rittmann, J.; Reinhard, M.; Penfold, T. J.; Chergui, M. Mapping of the Photoinduced Electron Traps in TiO_2 by Picosecond X-ray Absorption Spectroscopy. *Angew. Chem., Int. Ed.* **2014**, 53, 5858–5862.
- (24) Ismail, A. S. M.; Uemura, Y.; Park, S. H.; Kwon, S.; Kim, M.; Elnaggar, H.; Frati, F.; Niwa, Y.; Wadati, H.; Hirata, Y.; et al. Direct observation of the electronic states of photoexcited hematite with ultrafast 2p3d X-ray absorption spectroscopy and resonant inelastic X-ray scattering. *Phys. Chem. Chem. Phys.* **2020**, 22, 2685–2692.
- (25) Leshchev, D.; Harlang, T. C. B.; Fredin, L. A.; Khakhlulin, D.; Liu, Y.; Biasin, E.; Laursen, M. G.; Newby, G. E.; Haldrup, K.; Nielsen, M. M.; et al. Tracking the picosecond deactivation dynamics of a photoexcited iron carbene complex by time-resolved X-ray scattering. *Chem. Sci.* **2018**, 9, 405–414.
- (26) Carneiro, L. M.; Cushing, S. K.; Liu, C.; Su, Y.; Yang, P.; Alivisatos, A. P.; Leone, S. R. Excitation-wavelength-dependent small polaron trapping of photoexcited carriers in $\alpha\text{-Fe}_2\text{O}_3$. *Nat. Mater.* **2017**, 16, 819–825.
- (27) Vura-Weis, J.; Jiang, C.-M.; Liu, C.; Gao, H.; Lucas, J. M.; de Groot, F. M. F.; Yang, P.; Alivisatos, A. P.; Leone, S. R. Femtosecond $\text{M}_{2,3}$ -Edge Spectroscopy of Transition-Metal Oxides: Photoinduced Oxidation State Change in $\alpha\text{-Fe}_2\text{O}_3$. *J. Phys. Chem. Lett.* **2013**, 4, 3667–3671.
- (28) Uemura, Y.; Kido, D.; Koide, A.; Wakisaka, Y.; Niwa, Y.; Nozawa, S.; Ichiyanagi, K.; Fukaya, R.; Adachi, S.-i.; Katayama, T.; et al. Capturing local structure modulations of photoexcited BiVO_4 by ultrafast transient XAFS. *Chem. Commun.* **2017**, 53, 7314–7317.
- (29) Ravensbergen, J.; Abdi, F. F.; van Santen, J. H.; Frese, R.-N.; Dam, B.; van de Krol, R.; Kennis, J. T. M. Unraveling the Carrier Dynamics of BiVO_4 : A Femtosecond to Microsecond Transient Absorption Study. *J. Phys. Chem. C* **2014**, 118, 27793–27800.
- (30) Koide, A.; Uemura, Y.; Kido, D.; Wakisaka, Y.; Takakusagi, S.; Ohtani, B.; Niwa, Y.; Nozawa, S.; Ichiyanagi, K.; Fukaya, R.; et al. Photoinduced anisotropic distortion as the electron trapping site of tungsten trioxide by ultrafast W L_1 -edge X-ray absorption spectroscopy with full potential multiple scattering calculations. *Phys. Chem. Chem. Phys.* **2020**, 22, 2615–2621.
- (31) Uemura, Y.; Kido, D.; Wakisaka, Y.; Uehara, H.; Ohba, T.; Niwa, Y.; Nozawa, S.; Sato, T.; Ichiyanagi, K.; Fukaya, R.; et al. Dynamics of Photoelectrons and Structural Changes of Tungsten Trioxide Observed by Femtosecond Transient XAFS. *Angew. Chem., Int. Ed.* **2016**, 55, 1364–1367.
- (32) Uemura, Y.; Uehara, H.; Niwa, Y.; Nozawa, S.; Sato, T.; Adachi, S.; Ohtani, B.; Takakusagi, S.; Asakura, K. In Situ Picosecond XAFS Study of an Excited State of Tungsten Oxide. *Chem. Lett.* **2014**, 43, 977–979.
- (33) Penfold, T. J.; Szlachetko, J.; Santomauro, F. G.; Britz, A.; Gawelda, W.; Doumy, G.; March, A. M.; Southworth, S. H.; Rittmann, J.; Abela, R.; et al. Revealing hole trapping in zinc oxide nanoparticles by time-resolved X-ray spectroscopy. *Nat. Commun.* **2018**, 9, 478.
- (34) Santomauro, F. G.; Grilj, J.; Mewes, L.; Nedelcu, G.; Yakunin, S.; Rossi, T.; Capano, G.; Al Haddad, A.; Budarz, J.; Kinschel, D.; et al. Localized holes and delocalized electrons in photoexcited inorganic perovskites: Watching each atomic actor by picosecond X-ray absorption spectroscopy. *Struct. Dyn.* **2017**, 4, No. 044002.
- (35) Jang, H.; Kim, H.-D.; Kim, M.; Park, S. H.; Kwon, S.; Lee, J. Y.; Park, S.-Y.; Park, G.; Kim, S.; Hyun, H.; et al. Time-resolved resonant elastic soft x-ray scattering at Pohang Accelerator Laboratory X-ray Free Electron Laser. *Rev. Sci. Instrum.* **2020**, 91, No. 083904.
- (36) Park, S. H.; Kim, M.; Min, C.-K.; Eom, I.; Nam, I.; Lee, H.-S.; Kang, H.-S.; Kim, H.-D.; Jang, H. Y.; Kim, S.; et al. PAL-XFEL soft X-ray scientific instruments and X-ray optics: First commissioning results. *Rev. Sci. Instrum.* **2018**, 89, No. 055105.
- (37) Kang, H.-S.; Min, C.-K.; Heo, H.; Kim, C.; Yang, H.; Kim, G.; Nam, I.; Baek, S. Y.; Choi, H.-J.; Mun, G.; et al. Hard X-ray free-electron laser with femtosecond-scale timing jitter. *Nat. Photonics* **2017**, 11, 708–713.
- (38) Takubo, K.; Yamamoto, K.; Hirata, Y.; Yokoyama, Y.; Kubota, Y.; Yamamoto, S.; Yamamoto, S.; Matsuda, I.; Shin, S.; Seki, T.; et al. Capturing ultrafast magnetic dynamics by time-resolved soft x-ray magnetic circular dichroism. *Appl. Phys. Lett.* **2017**, 110, 162401.
- (39) Katayama, T.; Nozawa, S.; Umena, Y.; Lee, S.; Togashi, T.; Owada, S.; Yabashi, M. A versatile experimental system for tracking ultrafast chemical reactions with X-ray free-electron lasers. *Struct. Dyn.* **2019**, 6, No. 054302.
- (40) Katayama, T.; Hirano, T.; Morioka, Y.; Sano, Y.; Osaka, T.; Owada, S.; Togashi, T.; Yabashi, M. X-ray optics for advanced ultrafast pump-probe X-ray experiments at SACLA. This article will form part of a virtual special issue on X-ray free-electron lasers. *J. Synchrotron Radiat.* **2019**, 26, 333–338.
- (41) Katayama, T.; Owada, S.; Togashi, T.; Ogawa, K.; Karvinen, P.; Vartiainen, L.; Eronen, A.; David, C.; Sato, T.; Nakajima, K.; et al. A beam branching method for timing and spectral characterization of hard X-ray free-electron lasers. *Struct. Dyn.* **2016**, 3, No. 034301.
- (42) Tono, K.; Togashi, T.; Inubushi, Y.; Sato, T.; Katayama, T.; Ogawa, K.; Ohashi, H.; Kimura, H.; Takahashi, S.; Takeshita, K.; et al. Beamline, experimental stations and photon beam diagnostics for the hard x-ray free electron laser of SACLA. *New J. Phys.* **2013**, 15, No. 083035.
- (43) Ishikawa, T.; Aoyagi, H.; Asaka, T.; Asano, Y.; Azumi, N.; Bizen, T.; Ego, H.; Fukami, K.; Fukui, T.; Furukawa, Y.; et al. A compact X-ray free-electron laser emitting in the sub-ångström region. *Nat. Photonics* **2012**, 6, 540–544.
- (44) Huang, M. J.; Deng, G.; Chin, Y. Y.; Hu, Z.; Cheng, J. G.; Chou, F. C.; Conder, K.; Zhou, J. S.; Pi, T. W.; Goodenough, J. B.; et al. Determination of hole distribution in $\text{Sr}_{1-x}\text{Ca}_x\text{Cu}_{24}\text{O}_{41}$ using soft x-ray absorption spectroscopy at the Cu L_3 edge. *Phys. Rev. B: Condens. Matter Mater. Phys.* **2013**, 88, No. 014520.
- (45) Choudhury, D.; Rivero, P.; Meyers, D.; Liu, X.; Cao, Y.; Middey, S.; Whitaker, M. J.; Barraza-Lopez, S.; Freeland, J. W.; Greenblatt, M.; et al. Anomalous charge and negative-charge-transfer insulating state in cuprate chain compound KCuO_2 . *Phys. Rev. B: Condens. Matter Mater. Phys.* **2015**, 92, 201108.
- (46) Wang, Y.; Lany, S.; Ghanbaja, J.; Fagot-Revurat, Y.; Chen, Y. P.; Soldera, F.; Horwat, D.; Mücklich, F.; Pierson, J. F. Electronic structures of Cu_2O , Cu_4O_3 , and CuO : A joint experimental and theoretical study. *Phys. Rev. B: Condens. Matter Mater. Phys.* **2016**, 94, 245418.

- (47) Grioni, M.; Goedkoop, J. B.; Schoorl, R.; de Groot, F. M. F.; Fuggle, J. C.; Schäfers, F.; Koch, E. E.; Rossi, G.; Esteve, J. M.; Karnatak, R. C. Studies of copper valence states with Cu L_3 x-ray absorption spectroscopy. *Phys. Rev. B: Condens. Matter Mater. Phys.* **1989**, *39*, 1541–1545.
- (48) Ruiz-Fuertes, J.; Segura, A.; Rodríguez, F.; Errandonea, D.; Sanz-Ortiz, M. N. Anomalous High-Pressure Jahn-Teller Behavior in CuWO_4 . *Phys. Rev. Lett.* **2012**, *108*, 166402.
- (49) Ye, X.; Schmidt, J. E.; Wang, R.-P.; van Ravenhorst, I. K.; Oord, R.; Chen, T.; de Groot, F.; Meirer, F.; Weckhuysen, B. M. Deactivation of Cu-Exchanged Automotive-Emission NH_3 -SCR Catalysts Elucidated with Nanoscale Resolution Using Scanning Transmission X-ray Microscopy. *Angew. Chem.* **2020**, *132*, 15740–15747.
- (50) Czyżyk, M. T.; Sawatzky, G. A. Local-density functional and on-site correlations: The electronic structure of La_2CuO_4 and LaCuO_3 . *Phys. Rev. B: Condens. Matter Mater. Phys.* **1994**, *49*, 14211–14228.
- (51) Mohamed, M.; May, M. M.; Kanis, M.; Brützm, M.; Uecker, R.; van de Krol, R.; Janowitz, C.; Mulazzi, M. The electronic structure and the formation of polarons in Mo-doped BiVO_4 measured by angle-resolved photoemission spectroscopy. *RSC Adv.* **2019**, *9*, 15606–15614.
- (52) Wiktor, J.; Ambrosio, F.; Pasquarello, A. Role of Polarons in Water Splitting: The Case of BiVO_4 . *ACS Energy Lett.* **2018**, *3*, 1693–1697.